

Course Title: Advanced Powder Metallurgy

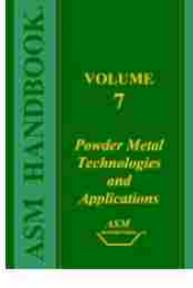
Weekly Hours: 2.0 semester hours per week (2 SWS)

Course schedule: Wednesdays, 13:00–15:00, Room 2.15

Lecturer: Prof. Dr. Maziyar AZADBEH 2nd semester of the academic year 1404-1403

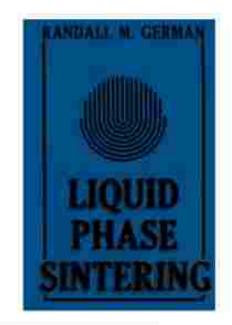
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J Mater Sci (2009) 44:1-50. DOI 10.1007/s10853-008-3008-0

REVIEW

Review: liquid phase sintering

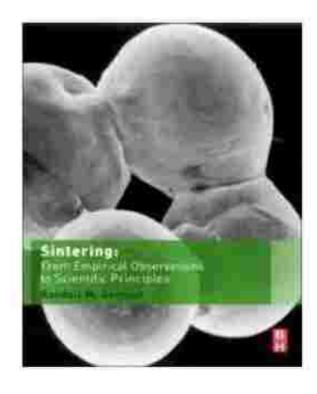
Randall M. German · Pavan Suri · Scong Jin Park

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Abstract Liquid phase sintering (LPS) is a process for forming high performance, multiple-phase components from powders. It involves sintering under conditions where solid grains coexist with a setting liquid. Many variants of LPS are applied to a wide range of engineering materials. Example applications for this technology are found in automobile engine connecting rods and high-speed metal

- Solid concentration in the matrix, m³/m³ or dimensionless
- Co. Grain connectivity, dimensionless
- Cxx Contiguity, dimensionless
- D Particle size, m (convenient units: um)
- D_i First eigenvalue of diameter of curvature, in (convenient units: μm)

5) RANDALL M. GERMAN, "SINTERING FROM EMPIRICAL OBSERVATIONS TO SCIENTIFIC PRINCIPLES", Elsevier, 2014





SINTERING FROM EMPIRICAL OBSERVATIONS TO SCIENTIFIC PRINCIPLES

RANDALL M. GERMAN

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First edition 2014

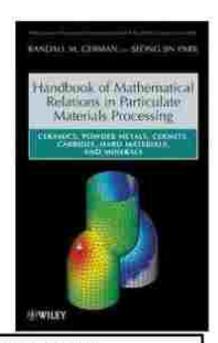
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6) RANDALL M. GERMAN, "MATHEMATICAL RELATIONS IN PARTICULATE MATERIALS PROCESSING", Published by John Wiley & Sons, Inc., 2008



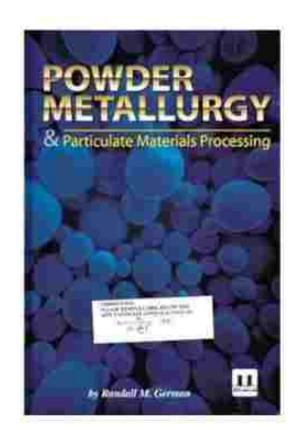
MATHEMATICAL RELATIONS IN PARTICULATE MATERIALS PROCESSING

Ceramics, Powder Metals, Cermets, Carbides, Hard Materials, and Minerals

RANDALL M. GERMAN SEONG JIN PARK



A JOHN WILEY & SONS, INC., PUBLICATION



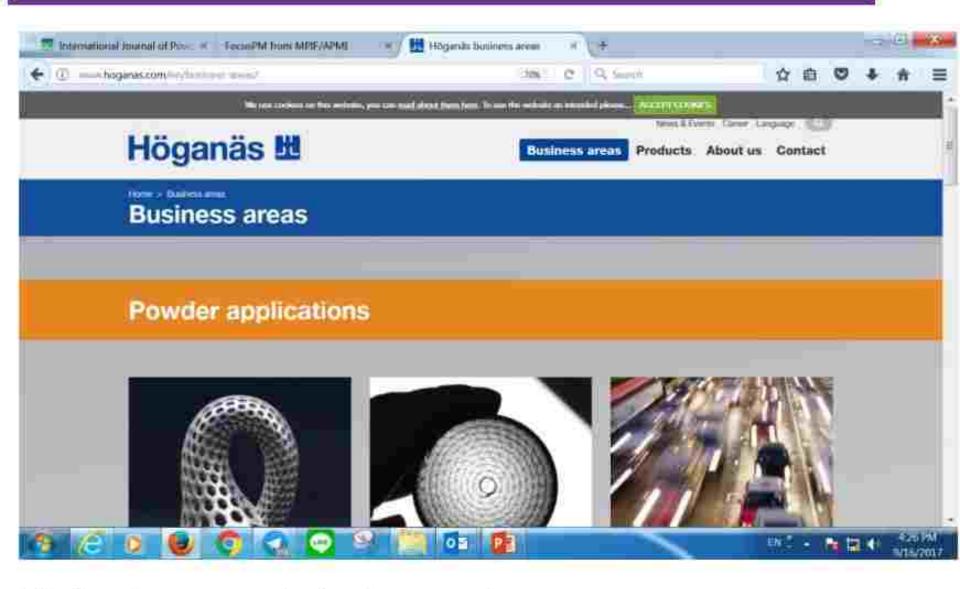


Supersolidus Liquid-Phase Sintering of Prealloyed Powders

RANDALL M. GERMAN

A model is derived for the sintering densification of prealloyed particles that form internal liquids when heated over the solides temperature. The model considers the powder size, composition, and microstructure, as well as the processing conditions of green density, heating rate, maximum temperature, hold time, and atmosphere. Internal liquid forms and spreads to create as interparticle capillary bond that induces densification during unitering. Densification is delayed until the particles achieve a mushy state due to grain boundary wetting by the internal liquid. This loss of rigidity and concomitant densification of the semisolid particles depends on the grain size and liquid quantity. Viscous flow is the assumed densification mechanism, where both viscousty and yield strength vary with the based content and particle microstructure. Densification profections are compared to exper-

7) Höganäs Handbook for sintered components



http://www.hoganas.com/en/business-areas/

8) W.Schatt and K.P.Wieters, "Powder Metallurgy, Processing and Materials", European Powder Metallurgy Association (EPMA), Shrewsbury, 1997.



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Powder Metallurgy - Processing and Materials



Author Schotl W. Wessen, X.P.

Farmet Hardwall

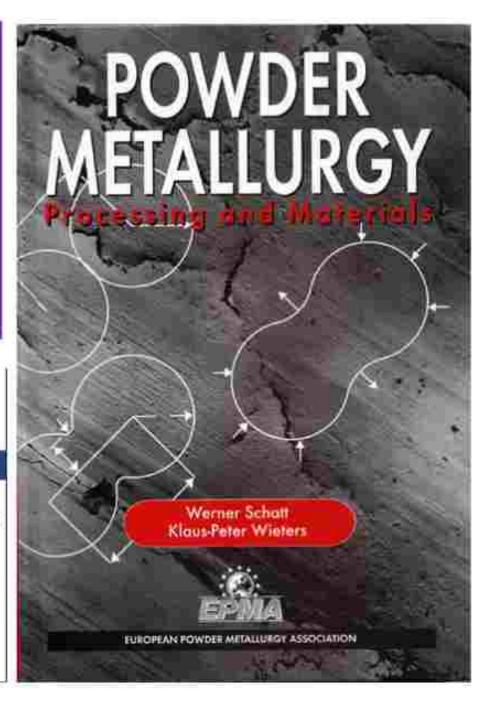
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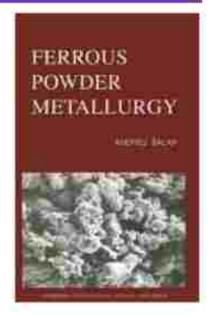
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Salak, Andrej.; Riecansky, V., Ferrous Powder Metallurgy, Cambridge International Science Publishing, 1995



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publisher: Cambridge International Science Publishing

isbn10 | asin: 1898326037 print isbn13: 9781898326038 ebook isbn13: 9780585165790

language: English

subject Powder metallurgy, Iron-metallurgy, Steel-

metallurgy.

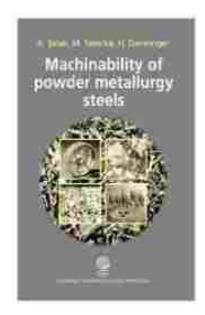
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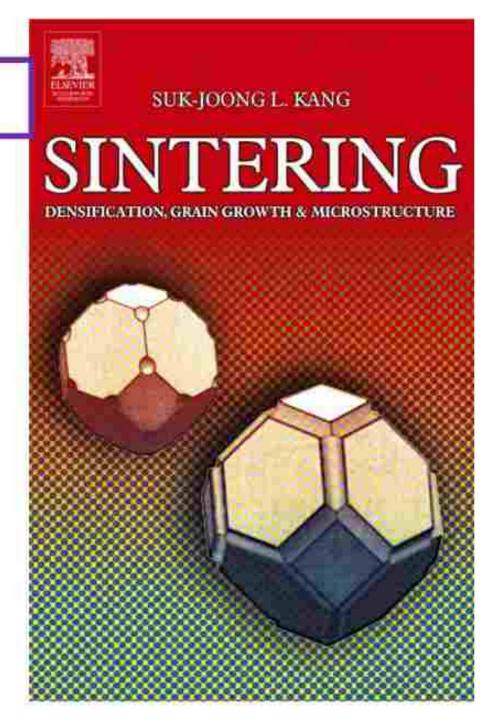
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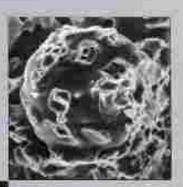
10) A.Salak, M.Selecka, H.Danninger, "Machinability of Powder Metalllurgy Steels", Cambridge International Science Publishing, 2005.



Suk-Joong L.Kang, "Sintering", Elsevier,
 2005



POWDER METALLURGY TECHNOLOGY





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Powder Testing Guide

Methods of measuring the physical properties of bulk powders

POWDER TESTING GUIDE

Methods of Measuring the Physical Properties of Bulk Powders

L. SVAROVSKY

School of Studies in Pander Technology, University of Boultons, UK



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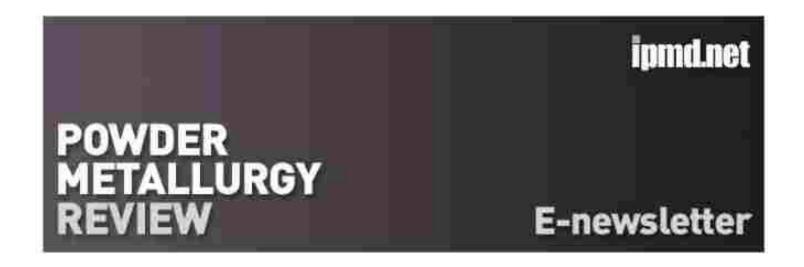
- گوردون داوسون،ترجمه دکتر علی حائریان اردکانی، "متالورژی پودر"،مرکز نشر دانشگاهی تهران،۱۳۷۲. عنوان اصلی:

Powder Metallurgy, the process and its products

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What is Powder Metallu

European Powder Metallurgy Association

Welcome to the European Powder Metallurgy Association (EPMA) Website. This website contains a wealth of knowledge on the Powder Metallurgy manufacturing process, as well as case studies, publications and details on industry related projects and events.

The Members Directory is one of the only online databases of Powder Metallurgy Companies and

Research Centres located around Europe

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INTRODUCTION TO ADDITIVE MANUFACTURI TECHNOLOGY





Introducing PM

About MPIF

PM Design Center

Directory of PM Fabricators & Suppliers

How to Reach the PM Market

Conferences & Exhibits

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Publications

Working in PM



Sunday, September 13, 2015

Welcome to the world's most comprehensive source for New to the world of powder powder metallurgy and metallurgy and particulate materials particulate materials? knowledge: the place START HERE to learn all to visit in order to keep about it. up with the latest advances in this exciting technology and industry.



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Metal Powder Industries Federation

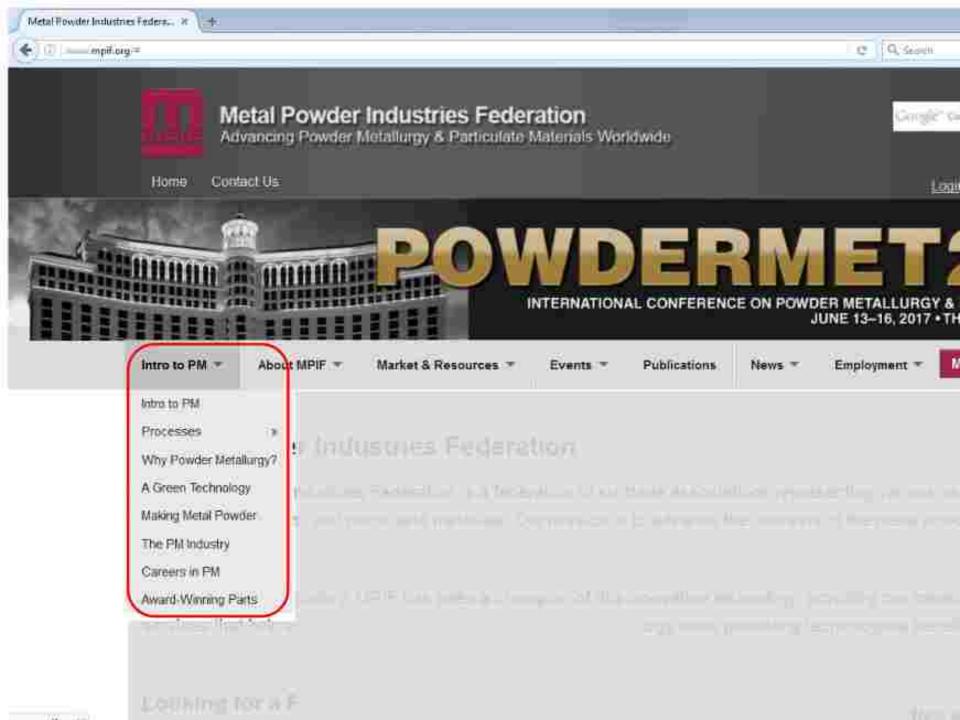
The Metal Powder Industries Federation is a federation of six trade associations representing various aspects of powder metallurgy (PM), metal powders, and particulate materials. Our mission is to advance the interests of the metal powder producing and consuming industries:

Throughout its long history, MPIF has been a champion of this innovative technology, providing our member companies with valuable services that help advance the art and science of powder metallurgy while promoting technological benefits to prospective end users.

Looking for a Powder Metallurgy Fabricator or

New and Upcoming





Introducing Powder Metallurgy

A complex planetary caller for a four-wheel drive torque transfer system. A helical gear and blades of stainless steel used in lapanoscopic surgical scissors. A manifold weighing over 6.5 tons used on an offshore oil platform. A steel connecting rod used in V-8 engines.

What do these diverse parts have in common? All were manufactured using a process called powder metallurgy (PM).

But what is powder metallurgy? Powder metallurgy is a metal-forming process performed by heating compacted metal powders to just below their melting points. Although the process has existed for more than 100 years, over the past quarter century it has become widely recognized as a superior way of producing high-quality parts for a veriety of important applications. This success is due to the advantages the process offers



over other metal forming technologies such as forging and metal cesting, advantages in material utilization, shape complexity, near-net-shape dimensional control, among others. These, in turn, contribute to sustainability, making

Powder metallurgy is an integral part of our lives.

To learn more about this metal forming process vestors:

Powder Metallurgy Touches Your Life, Part 1



Powers Metallurgy. The Frederics Metal Forming Solution showcases the fabrication capabilities of the various secretary representations of the various secretary (PM). Built on the Cheme. "Every Cay, in some way, PM touches your life," the two-part video uses docens of examples of actual components manufactured for many different applications to illustrate the benefits PM offers parts designers and engineers.

Powder Metallurgy Touches Your Life, Part 2



Metal Injection Molding Touching Your Life, Part 1



This video, in two parts, showcases the fabrication capabilities of the various technologies incom collectively as precise metallicity (PM), with particular focus on metal injection motions. Part 1 of the program uses occurs of examples of accusal components manufactured for many different applications to illustrate the benefits all the various PM technologies offer parts designers and engineers. Part 2 of the

program describes the medal injection misking process, providing design engineers with all the facts needed to understand why if they're designing an intricate metal part, they should think of MIM from the start.

Metal Injection Molding Touching Your Life, Part 2





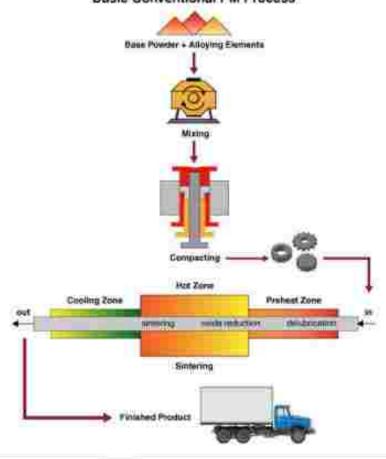


Conventional Powder Metallurgy Process

The powder metalurgy (FIII) process, and known as press-and-ainter, depicted in the diagram below, consists of moving elemental or alloy powders, compacting the mixture in a die, and then sintering, or heating, the resultant shapes in an atmosphere-controlled furnace to bond the particles midalurgically.

stast powder metallurgy parts weigh less than 5 pounds (2.27 kg), although parts weighing as much as 35 pounds (15.85 kg) can be fabricated in conventional PM equipment. While many of the early PM parts, such as bushings and bearings, were very simple shapes, today's sophisticated powder metallurgy process produces components with complex contours and multiple levels, and does so oute economically.

Basic Conventional PM Process



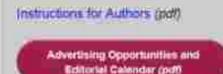


International Journal of Powder Metallurgy



The International Journal of Powder Metallurgy is published four times a year, bringing you the latest news in the powder metallurgy (PM) and particulate materials industries.

The Journal embraces a wide range of materials and processes including classical "press-and-sinter" powder metallurgy, metal injection molding, metal additive manufacturing, and advanced particulate materials. Its editorial coverage features distinct aspects of the PM industry such as:



















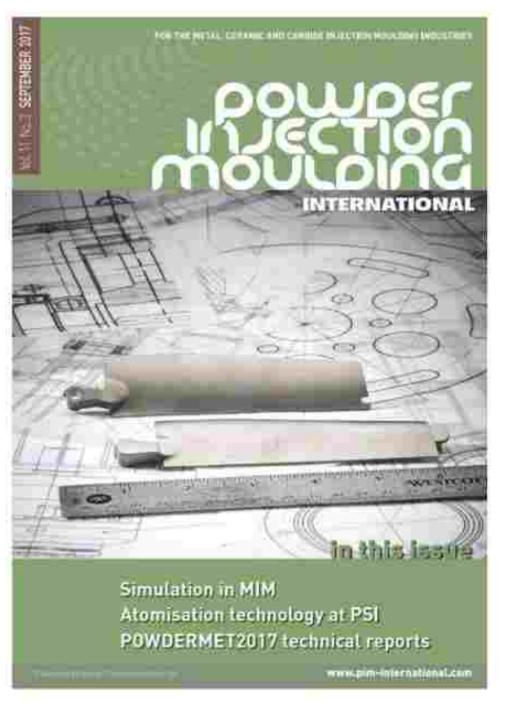


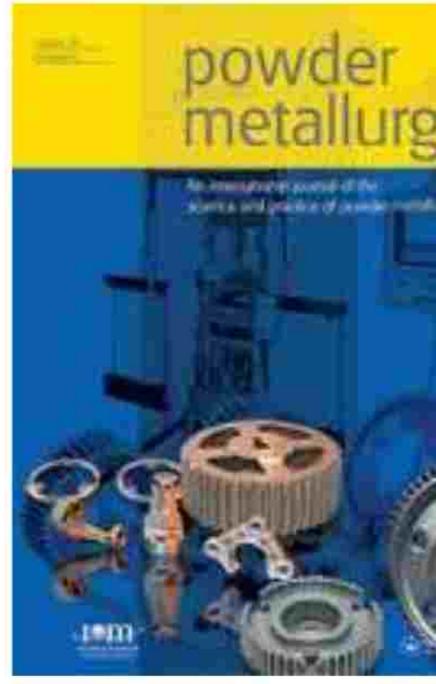












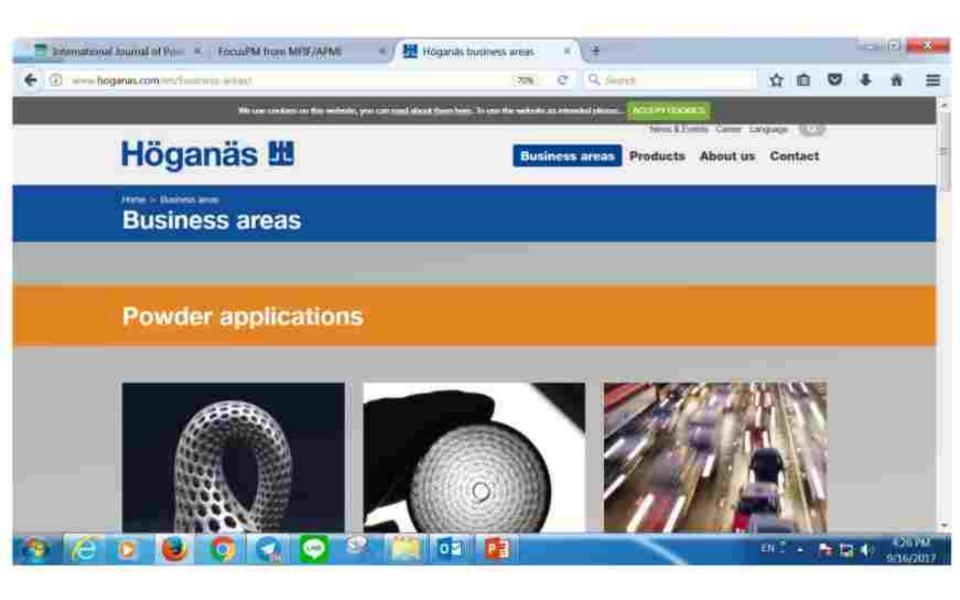
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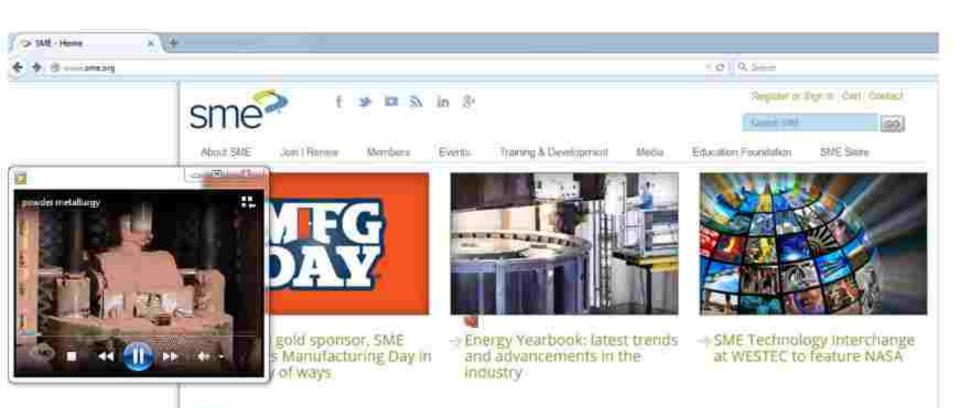
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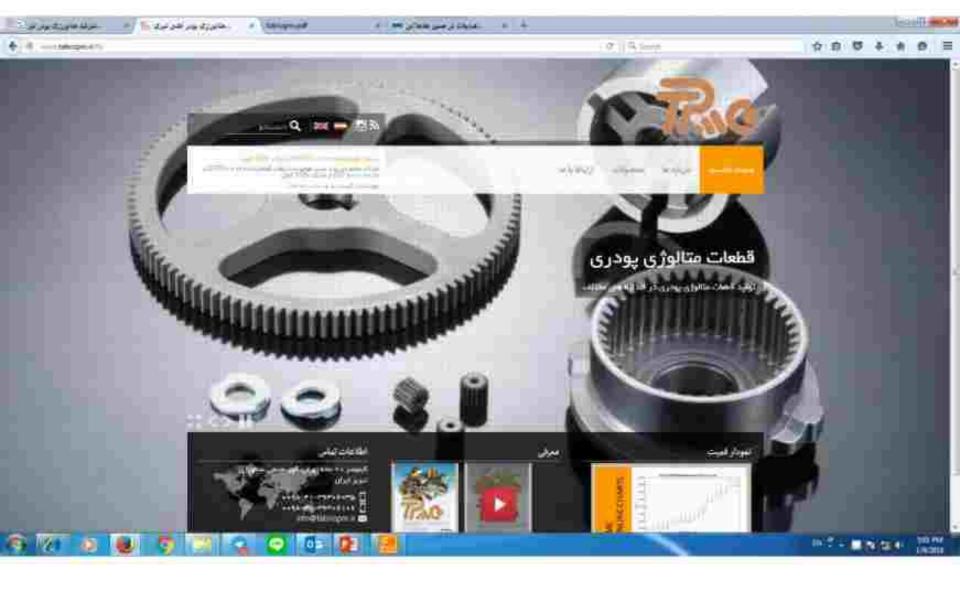


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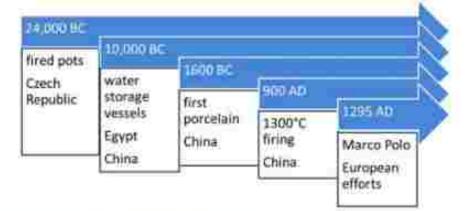
شرکت فر آورده های متالورژی پودری سهند













early **Sintered pots** were not sintered at high temperatures, so they were weak and rarely survived. They leaked liquids because of their porosity.

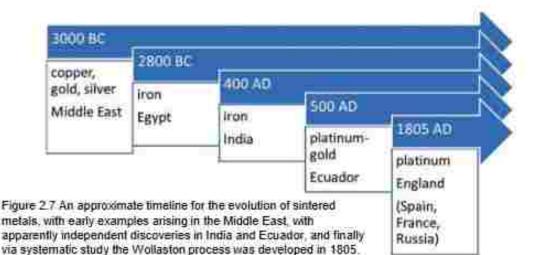
By approximately 10,000 BC, fired clay vessels were used for water storage, indicating techniques for sealing surface pores had been mastered.



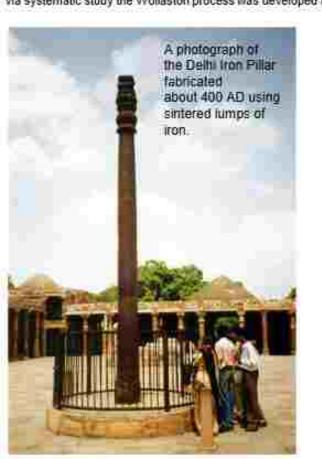
Figure 2.5 Early Chinese porcelain production excelled in reaching high sintering temperatures by use of sloped dragon kiln design, sketched in this early illustration.



European porcelain eventually found composition formulations and sintering techniques capable of forming high value products, initially near Dresden.



The first of these are from approximately 3000 BC [4,5]. Most of these easily reduced metals were sintered long ago.

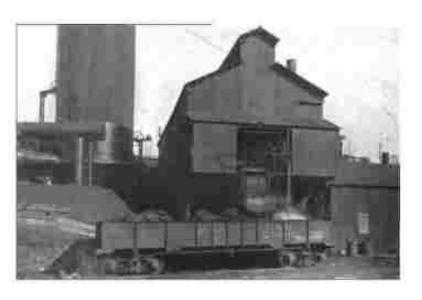


Platinum Crucibles



Figure 2.10 Photograph of an early sintered goldplatinum decorative medailion.







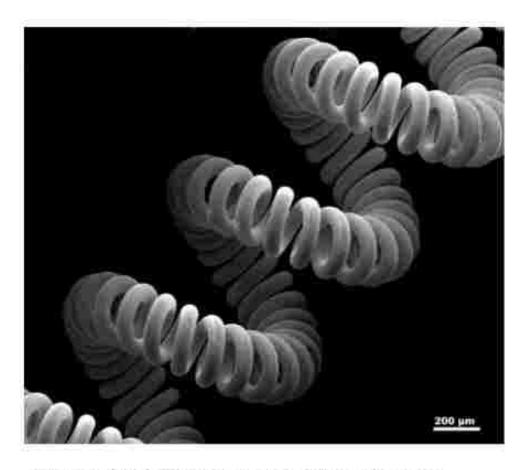


Figure 2.16 This is a scanning electron micrograph of a coiled tungsten lamp filament.





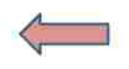
متالورژی پودر علم نوینی است که توسعه آن از مسیری بسیارابهام انگیز گذشته است.

متالورژی پودرشیوه ای است نوین برای شکل دهی فلزات. متالورژی پودر نه یک هنر،بلکه تکنولوژی موفقی است که در خط مقدم تکوین مفاهیم تازه مهندسی مواد و شیوه های نوین شکل دهی فلزات جای دارد. فرایند تبدیل پودربه قطعه را می توان از جنبه های تکنولوژیکی و یا علمی مورد توجه قرار داد.

در حالت اول (جنبه تکنولوژیکی) تکیه بر روش تولید مواد و قطعات خاص است.

در حالیکه از جنبه علمی،متالورژی پودر به منزله ترکیبی از قوانین پایه ای ترمودینامیک، مکانیک و سینتیک تلقی می گردد

در این تکنولوژی نه تنها شیمی ماده بلکه موارد زیر قابل کنترل می باشد :



پارامترهای عملیات حرارتی

پارامترهای ریزساختاری

توزیع فازها و ریزسازهای قطعه (micro constituents)

از جمله تخلخل

متالورژی پودر که معمولا با حروف اول واژه های انگلیسی آن یعنی PM یا Powder Metallurgy) P/M)

مشخص می شود را می توان

"تولید فراورده های مفید از پودر فلز ،بدون گذر از حالت مذاب" تعریف کرد.

در این مبحث فرایندهای

تولید پودر و رفتار پودر در مراحل فشردن بعدی تاثیر گذارند که به نحو چشمگیری کل فرایند تولید را تحت تاثیر قرار می دهد.

فعالیت های متالورژی پودر

تعریف پودر:

در این مرحله بجاست که سوال کنیم "منظور از پودرچیست؟"

یک پاسخ اولیه ممکن است این باشد که البته همه می دانند پودر چیست؟

اما این سوال واقعا به این سادگی نیست. آیا مجموعه ساچمه هایی به قطر 5mm پودر است؟ البته که نیست؛پس مرز کجاست؟

تعریف پودر:

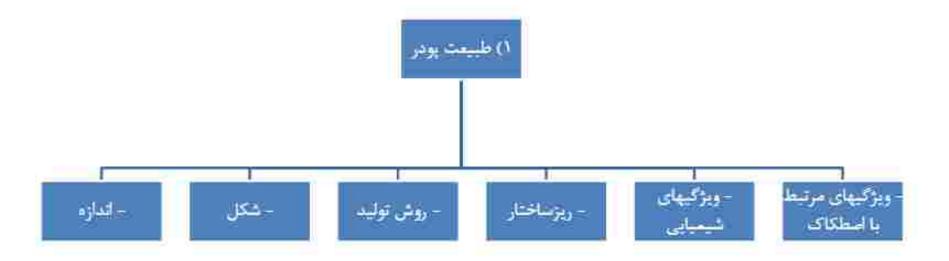
یک تعریف قابل قبول:

پودر عبارتست از دانه های ریزیک جامد که بزرگترین بعد آنها از 1mm کوچکتر باشد. (ISO 3252)

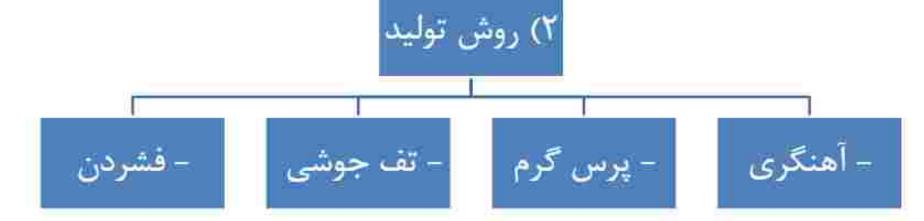
در بیشتر موارد پودرها فلزی می باشند ولی در مواردی با فازهای دیگر هم چون سرامیک ها و پلی مرها نیز مخلوط می شوند.

یکی از ویژگی های مهم پودر، بزرگی $\frac{Surface\ Area}{Volume}$ می باشد.

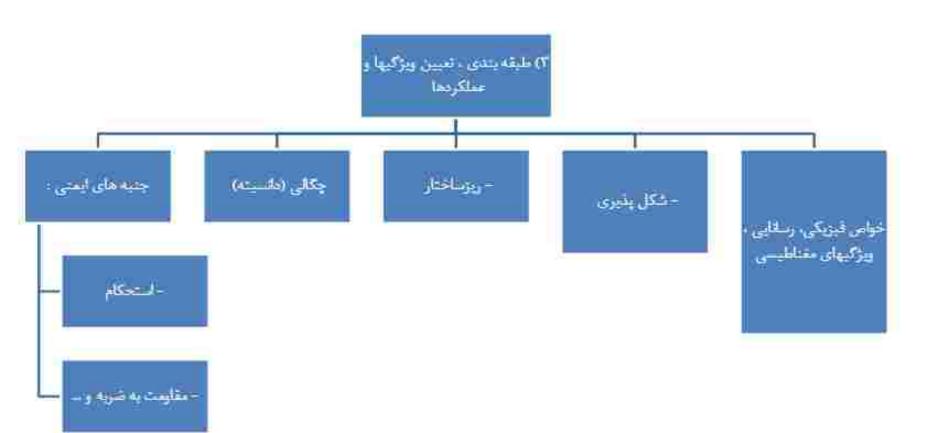
تقسیم بندی فعالیت های متالورژی پودر (تکنولوژی پودر)



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تقسیم بندی فعالیت های
متالورژی پودر
(تکنولوژی پودر)
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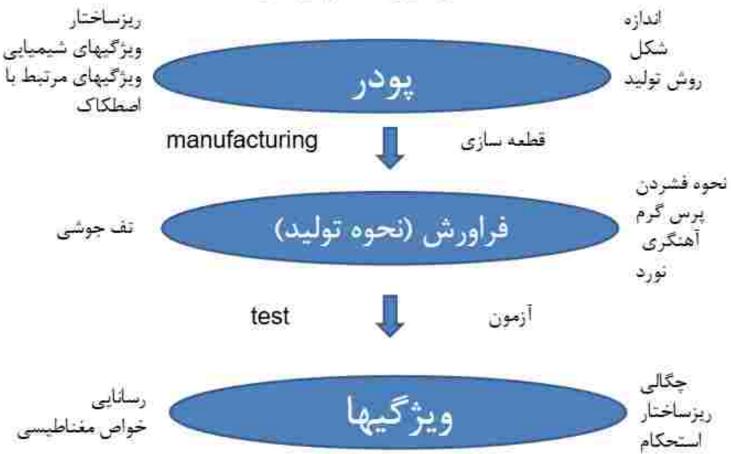


تقسیم بندی فعالیت های متالورژی پودر (تکنولوژی پودر)



•۴)بسته بندی و انتقال آن

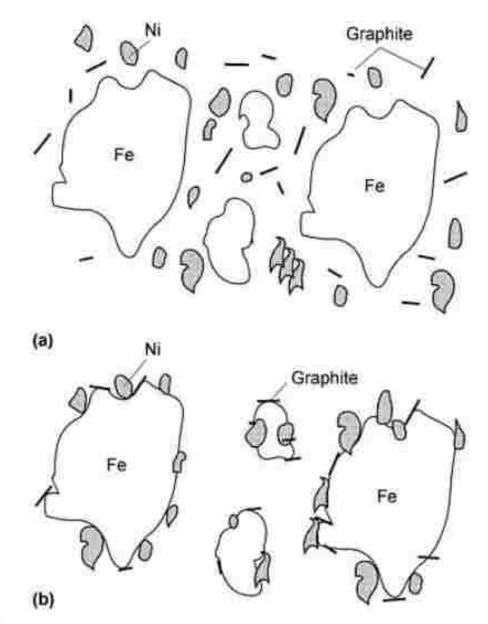
تکنولوژی پودر



Introduction

Powder Metallurgy Process

| Date | Description | Origin |
|-----------------|---|------------------------|
| 3000 BC | "Sponge iron" for making tools | Egypt, Africa, India |
| AD 1200 | Cementing platimum grains | South America |
| 1781 | Fusible plannum-arsenic alloy | France, Germany |
| 1790 | Production of platinum-arsenic chemical vessels commercially | France |
| 1822 | Platinum powder formed into solid ingot | France |
| 1826 | High-temperature sintering of platinum powder compacts on a commercial basis | Russia |
| 1829 | Wollaston method of producing compact platinum from platinum sponge (basis of modern P/M technique) | England |
| 1830 | Sintering compacts of various metals | Europe |
| 1859 | Platinum fusion process | |
| 1870 | Patent for bearing materials made from metal powders self-lubricating bearings | United States |
| 1878-1900 | Incandescent lamp filaments | United States |
| 1915-1930 | Cemented carbides | Germany |
| Early 1900s | Composite metals | United States |
| | Porous metals and metallic filters | United States |
| 1920s | Self-lubricating bearings (used commercially) | United States |
| 1940s | Iron powder technology | Central Europe |
| 1950s and 1960s | P/M wrought and dispersion-strengthened products, including P/M forgings | United States |
| 1970s | Hot isostatic pressing, P/M tool steels and superplastic superalloys | United States |
| 1980s | Rapid solidification and powder injection molding technology | United States |
| 1990s | Intermetallics, metal-matrix composites, spray forming, nanoscale powders and warm compaction | United States, England |



Alloy distribution.

- (a) Regular premix.
- (b) Segregation-free premix with powder binder

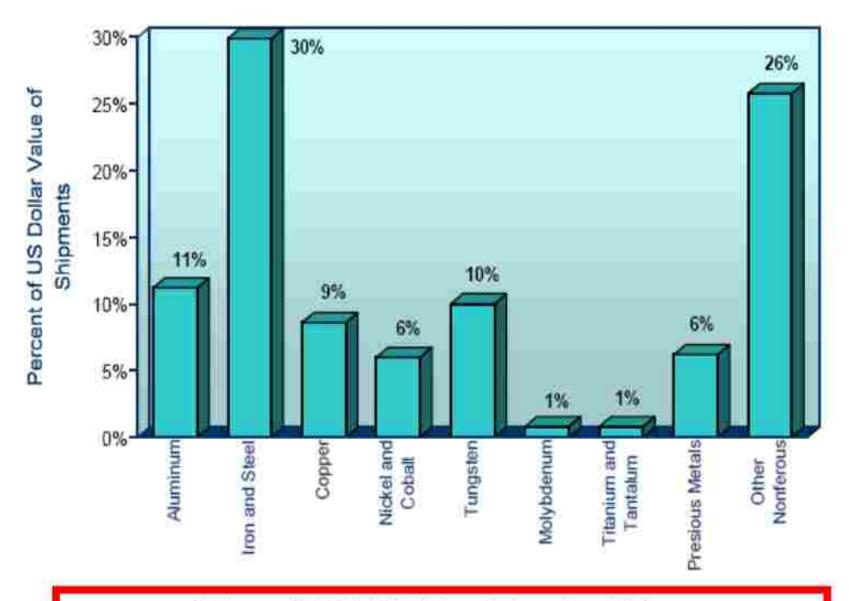


Fig. Value of 1997 US Metal Powder Shipments

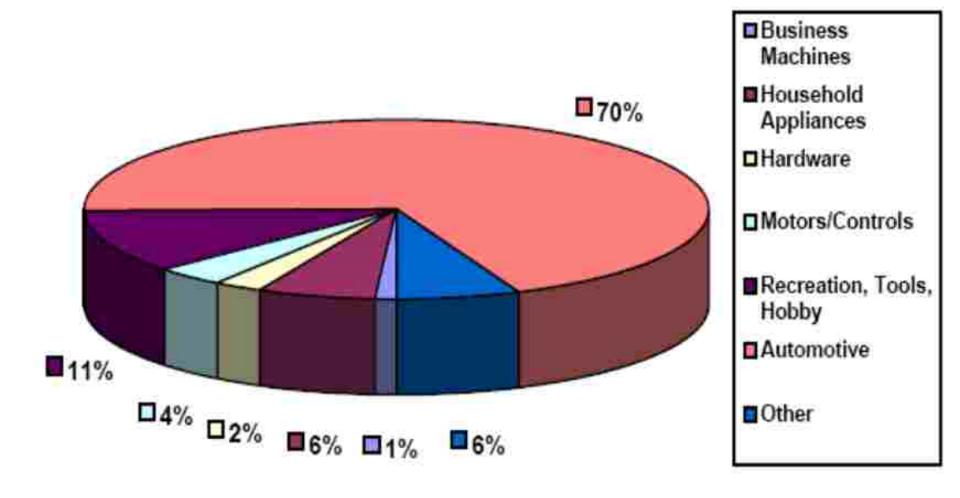


Fig. End Use of Application for Metal Powder

















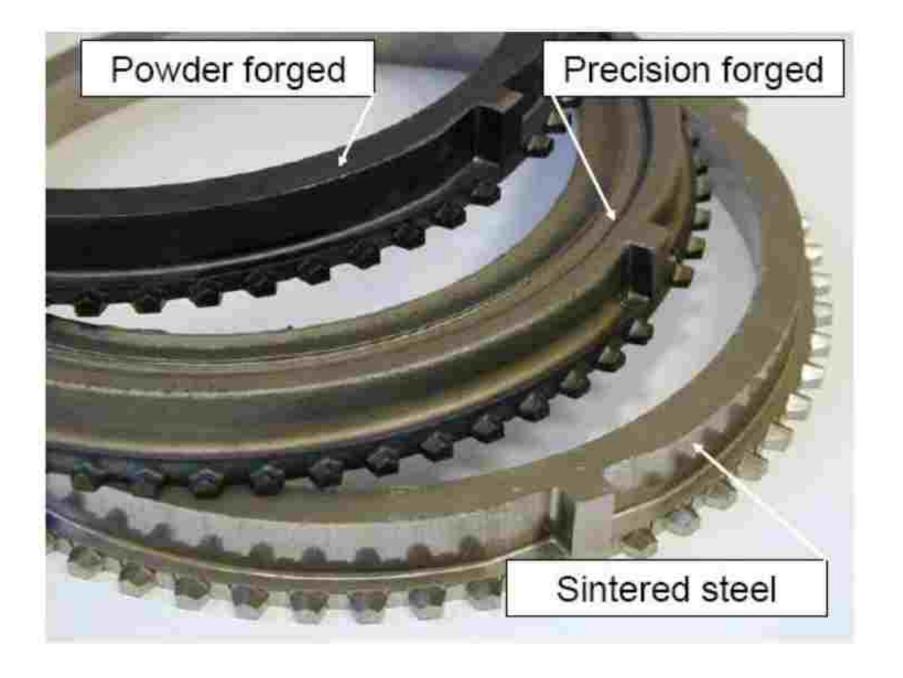




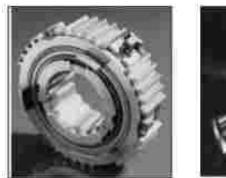








متالورژي پودر در صنعت خودرو















خالورزی بودر در صحت خودروسازی











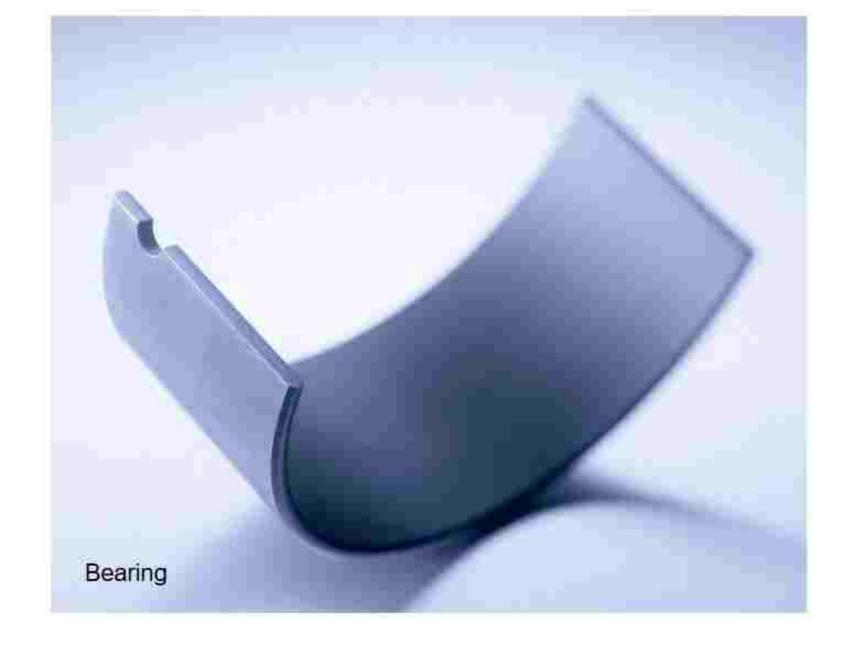






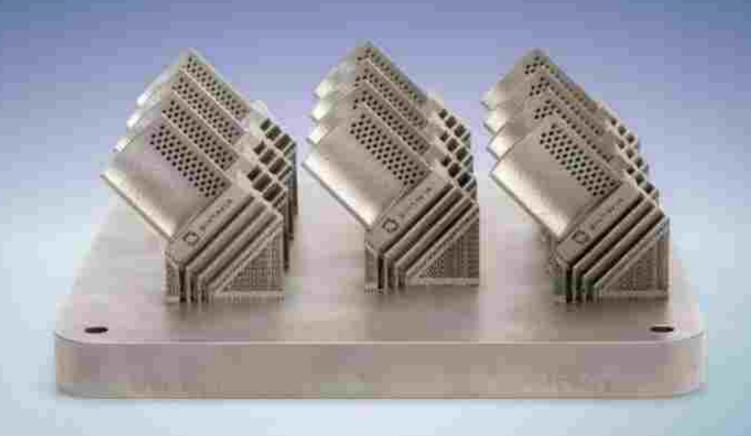


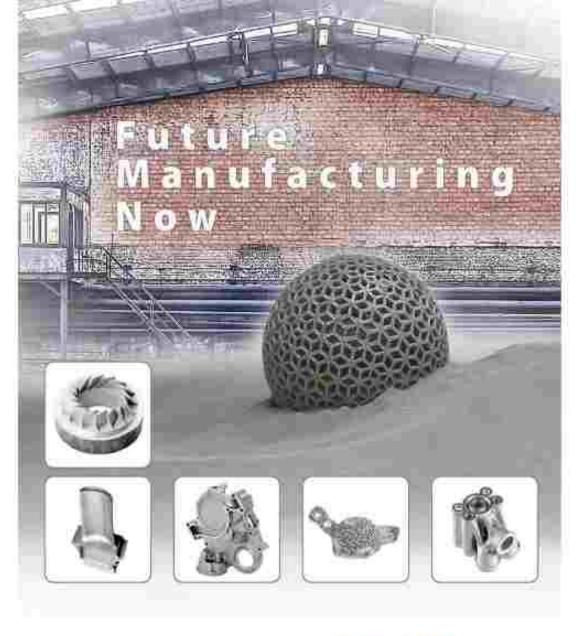
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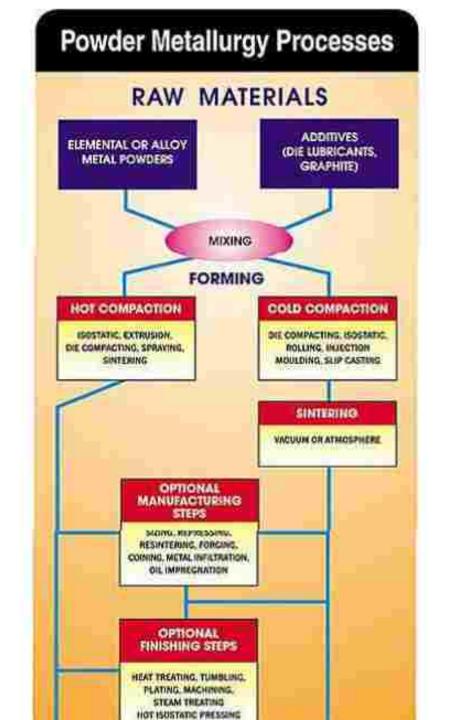
200 College Place Cold - Proceeding New Jorney (2004) 0007 U.S.A. TLL 1000 410 7700 - New York 197 4000 - Firmer South State (197 By producing parts with a homogeneous structure the PM process enables manufacturers to make products that are <u>more consistent</u> and <u>predictable in their behaviour</u> across a wide range of applications.

Advantages:

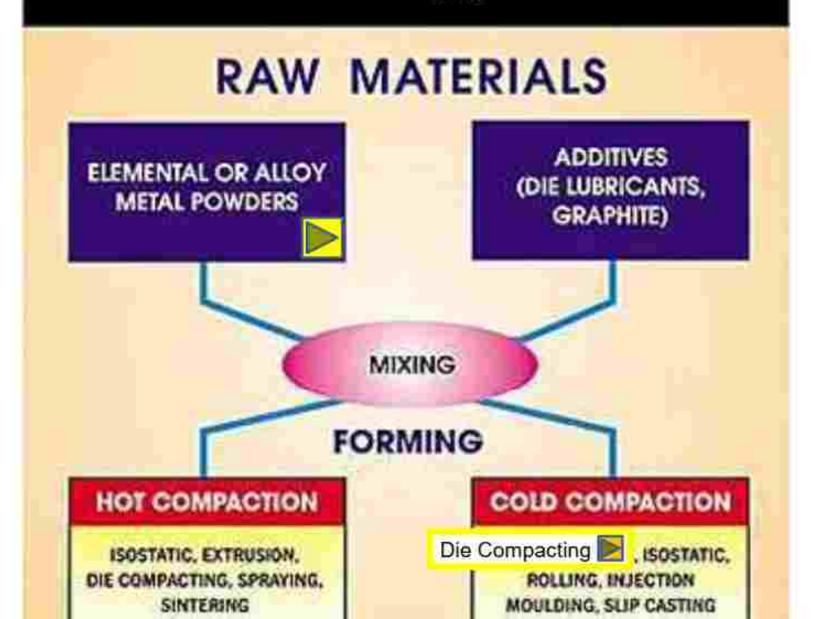
- >Structural pieces with complex shapes
- Controlled Porosity
- Controlled performance
- Good performance in stress and absorbing of vibration
- Special properties such as hardness and wear resistance
- Great precision and good surface finish
- Large series of pieces with narrow tolerances



The high precision forming capability of PM generates components with near net shape, intricate features and good dimensional precision pieces are often finished without the need of machining.



Powder Metallurgy Processes



Alloying Methods of Iron Powders

Completely Alloyed Powder







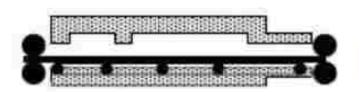
Mixed Alloyed Powder



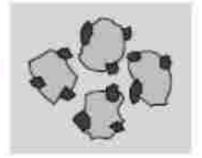




Partially Alloyed Powder

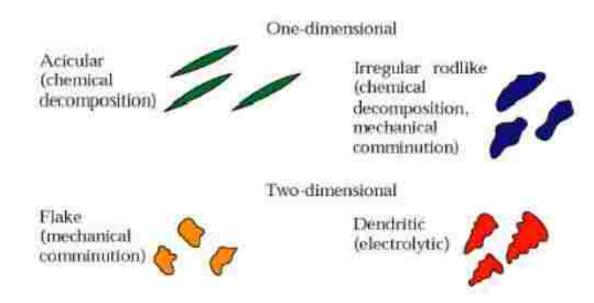








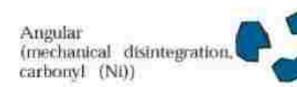
Particle Shapes in Metal Powders



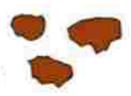
Three-dimensional

Spherical (atomization, carbonyl (Fe), precipitation from a liquid)

Rounded (atomization, chemical decomposition)



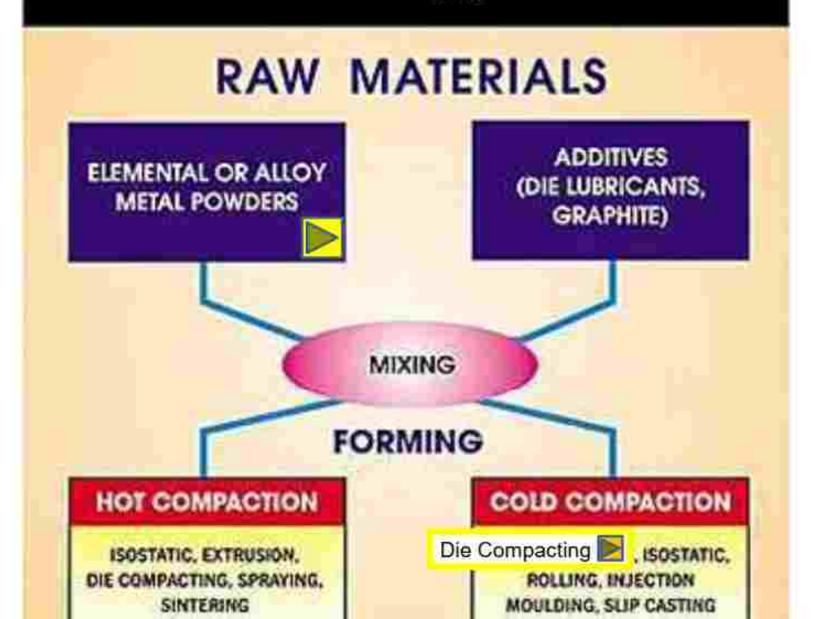
Irregular (atomization, chemical decomposition)



Porous (reduction of oxides)



Powder Metallurgy Processes



Material:



Fe-Cr-Mo-C (Fe - 1.5% Cr - 0.2% Mo)+ 0.6% C



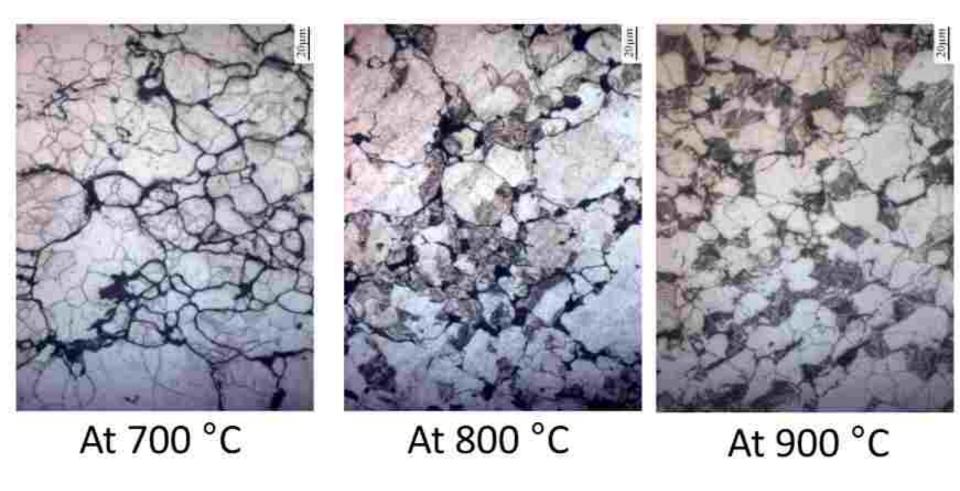
| Apparent density | 2.85 g/cm ³ |
|------------------|------------------------|
| low | 26 sec/50g |
| rticle size | 45-150 µm |
| Chemi | cal analysis, % |
| C | < 0.01 |
| Chromium | 1.50 |
| Molybdenum | 0.20 |



Fe-Mo-C

(Fe - 1.5% Mo)+ 0.6% C

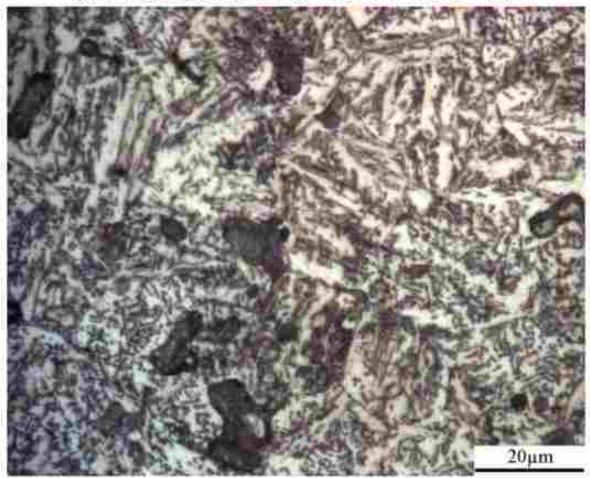
| And the same of th | Properties of Fe-Mo powder | | |
|--|----------------------------|-----------------------|--|
| | Apparent density | 3.1 g/cm ⁵ | |
| | Flow | 25 sec/50g | |
| | Particle size | 45-150 µm | |
| | Chemical analysis, % | | |
| | C | <0.01 | |
| | Molybdemum | 1,50 | |



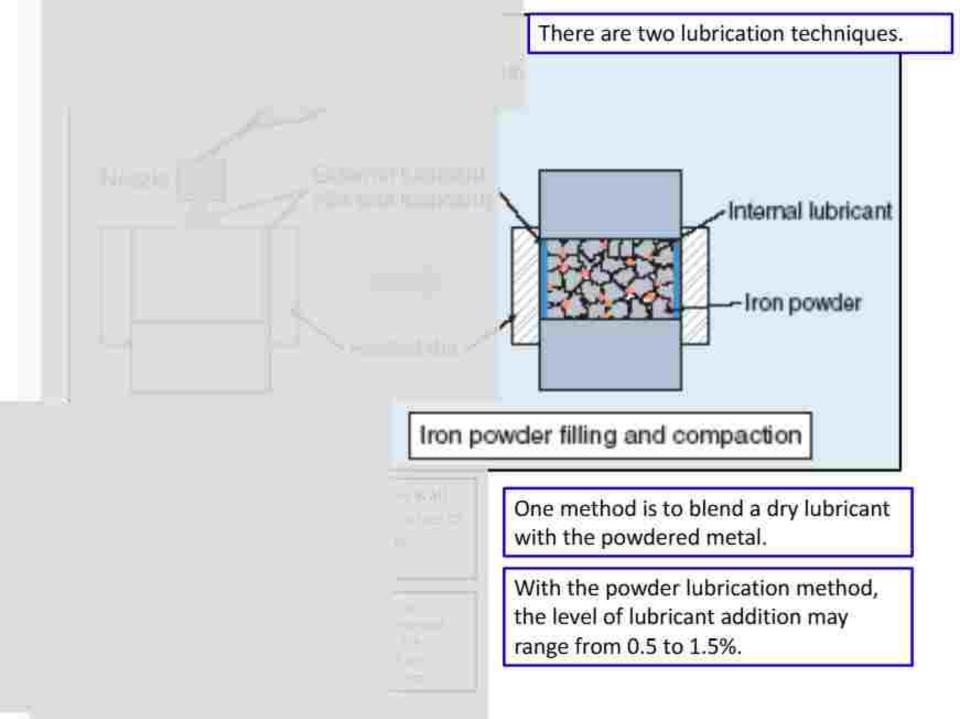
Microstructure of Fe-Mo-C,
compacted at 600 MPa and

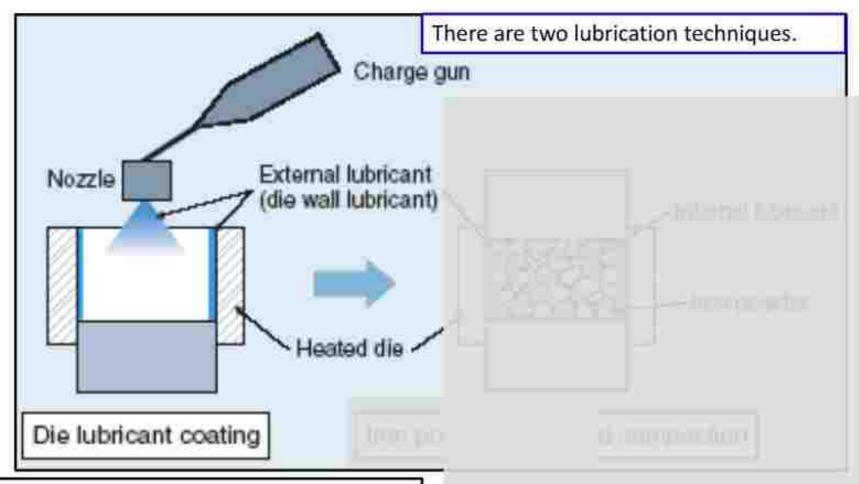
1st sintered at the indicated temperature for 30 min, and
repressed at 600 MPa. (As repressed)

Sintered Fe-Cr-Mo-C



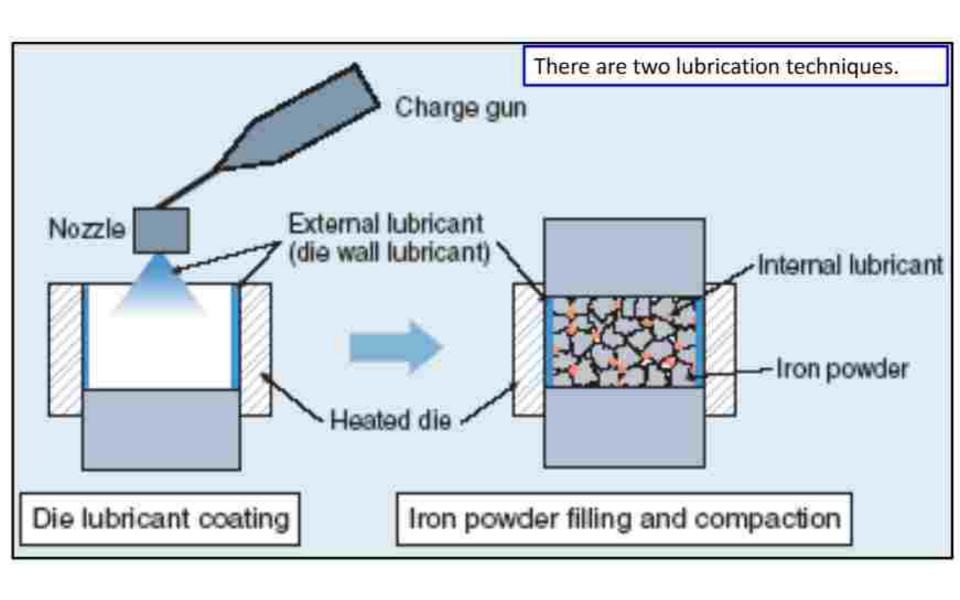
1st sintering Temperature=800°C





The other method, commonly referred to as die wall lubrication, is to lubricate the die walls and punches of the compacting tooling prior to introducing the powder metal into the die cavity.

With die wall lubrication, the solid lubricant (for example, zinc stearate, 100 g) is mixed with a volatile organic liquid (for example, methylchloroform, 1 L) and is either painted or sprayed on the tooling. The organic liquid evaporates, leaving a thin film of dry lubricant on the working surfaces of the die cavity and punches.



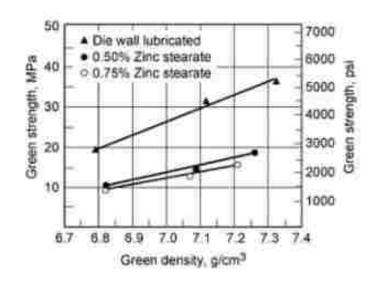


Fig. 9 Effect of lubrication method on green strength. Source: Ref 1

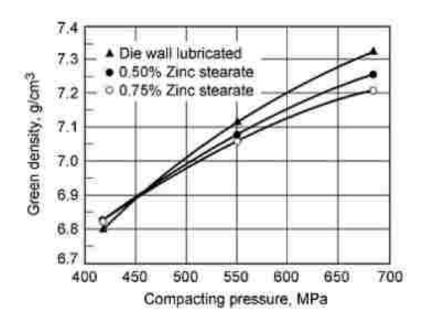


Fig. 10 Effect of lubrication on green density. Source: Ref 1

Lubricants. The effect of lubricants on compressibility depends on the lubricant and the compaction pressures. Some lubricants perform better than others, but generally lubricants cause a reduction in green strength (Fig. 9).

The reduction in compressibility from lubricants is more pronounced at higher compaction pressures (Fig. 10). At higher compaction pressures, lubricants reduce the green density because it occupies the available porosity.



There are two lubrication techniques. One method is to blend a dry lubricant with the powdered metal.

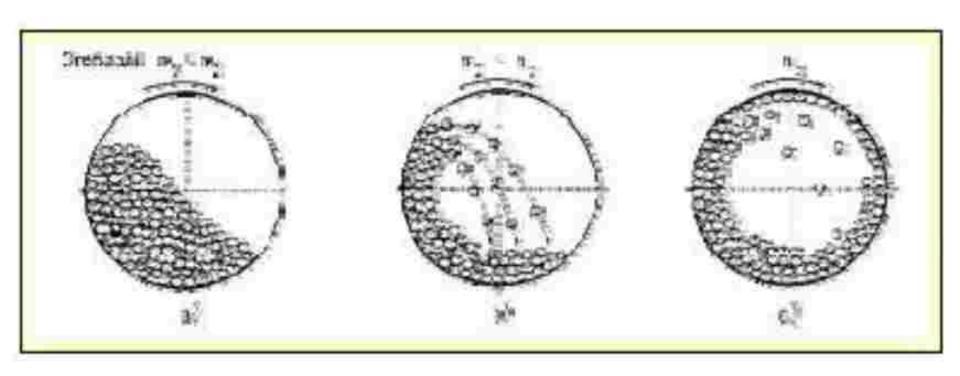
With the powder lubrication method, the level of lubricant addition may range from 0.5 to 1.5%.

The other method, commonly referred to as die wall lubrication, is to lubricate the die walls and punches of the compacting tooling prior to introducing the powder metal into the die cavity.

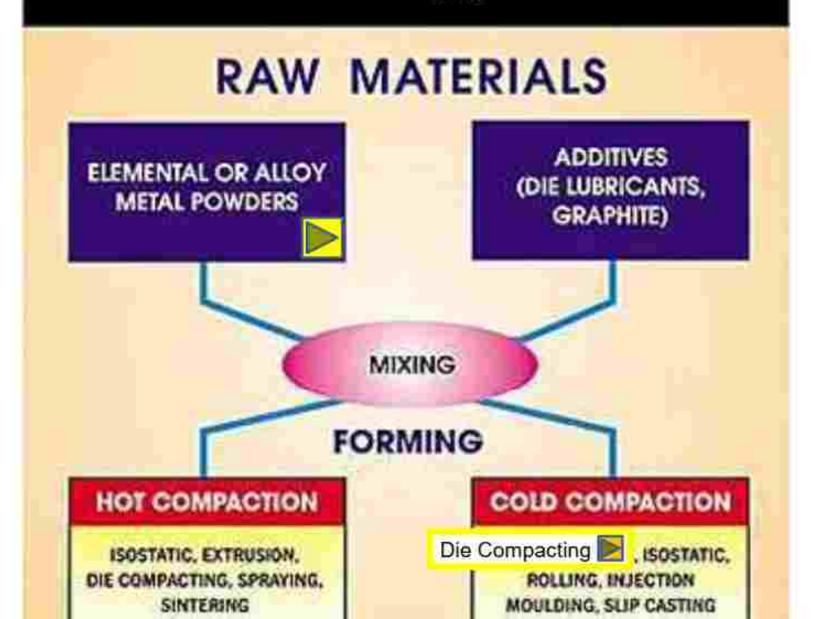
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Powder Metallurgy Processes

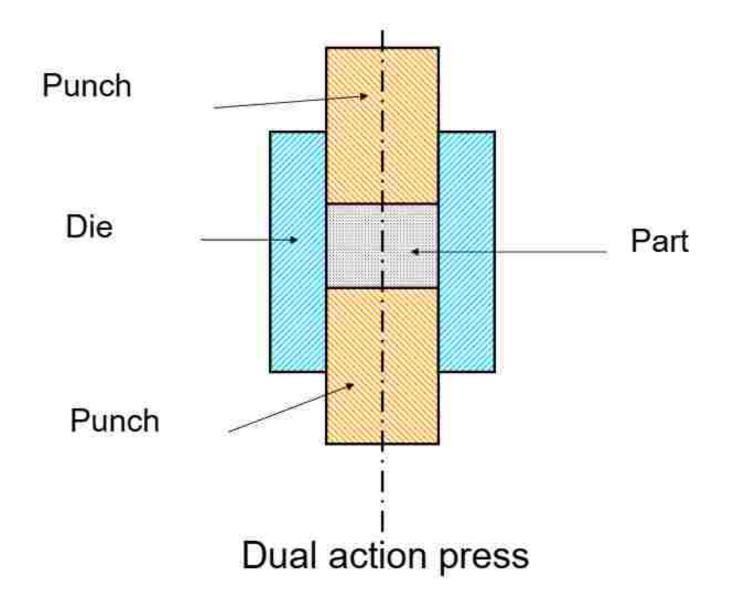
RAW MATERIALS **ADDITIVES** ELEMENTAL OR ALLOY (DIE LUBRICANTS, **METAL POWDERS** GRAPHITE) MIXING FORMING COLD COMPACTION HOT COMPACTION Die Compacting , Isostatic. ISOSTATIC, EXTRUSION. DIE COMPACTING, SPRAYING, ROLLING, INJECTION MOULDING, SLIP CASTING SINTERING

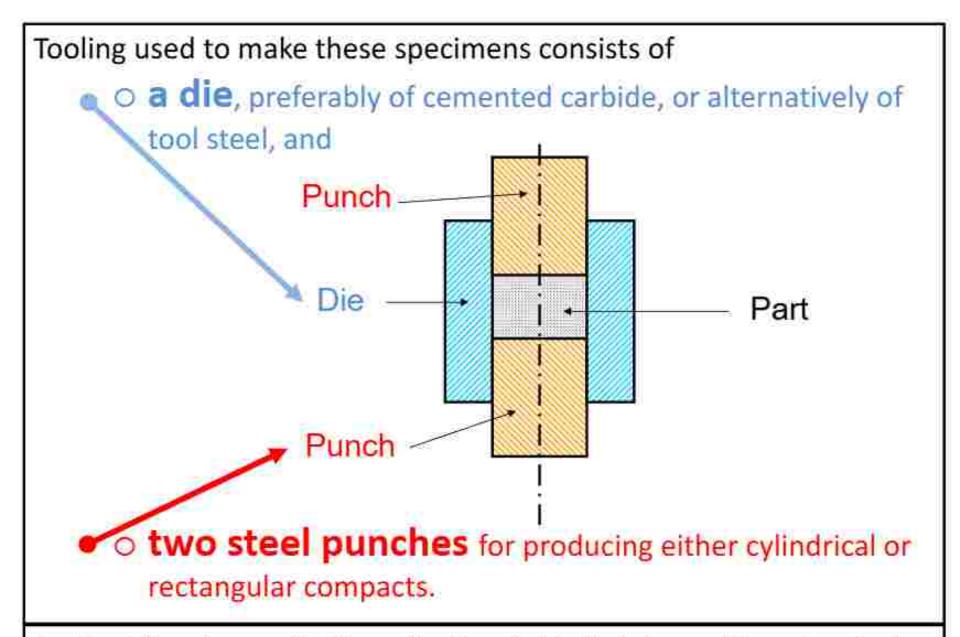


Powder Metallurgy Processes



Powder Pressing

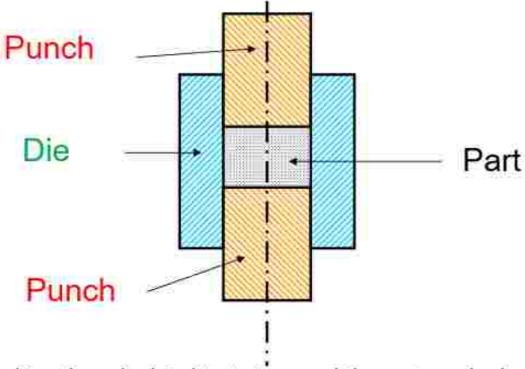




Exact tool dimensions required for making the cylindrical test piece and the rectangular bar are described in MPIF 45 and ISO 3927.

Tooling used to make these specimens consists of

- a die, preferably of cemented carbide, or alternatively of tool steel, and
- two steel punches for producing either cylindrical or rectangular compacts.



Exact tool dimensions required for making the cylindrical test piece and the rectangular bar are described in MPIF 45 and ISO 3927.

Compacting

 Loose powder is compacted and densified into a shape, known as green compact

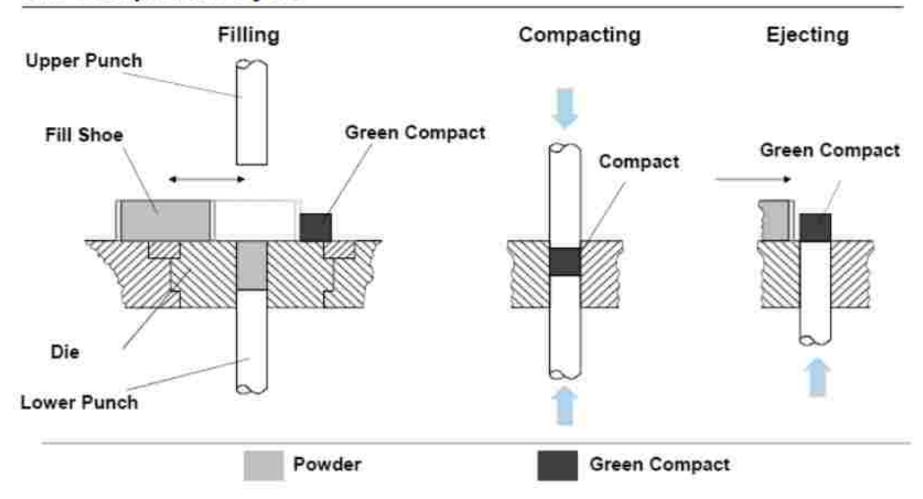
Most compacting is done with mechanical presses and

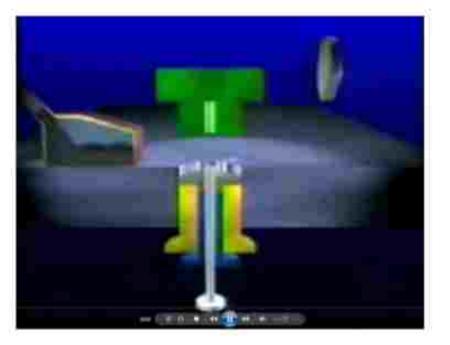
rigid tools

| TABLE 18-1 Typical Compacting Pressu | ures for Various Applications |
|--------------------------------------|-------------------------------|
|--------------------------------------|-------------------------------|

| | Compaction Pressures | |
|---|----------------------|----------|
| Application | tons/in.2 | Mpa |
| Porous metals and filters | 3-5 | 40-70 |
| Refractory metals and carbides | 5-15 | 70-200 |
| Porous bearings | 10-25 | 146-350 |
| Machine parts (medium-density iron & steel) | 20-50 | 275-690 |
| High-density copper and aluminum parts | 18-20 | 250-275 |
| High-density iron and steel parts | 50-120 | 690-1650 |

The Compaction Cycle





Compressibility Testing

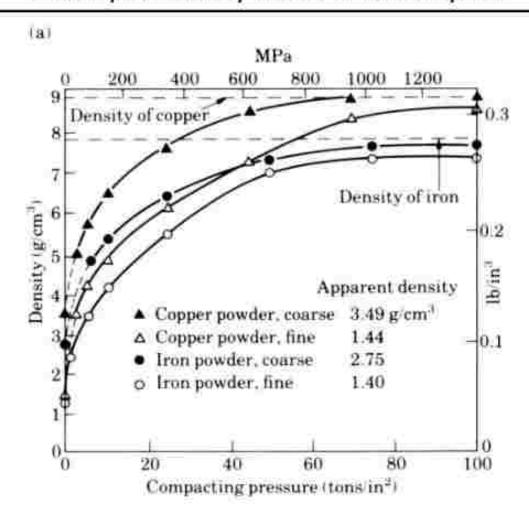
Standard test methods for compressibility have been issued as:

- ASTM B331, "Compressibility of Metal Powder in Uniaxial Compaction," by the American Society for Testing and Materials
- MPIF 45, "Determination of Compressibility of Metal Powders," by the Metal Powder Industries Federation
- ISO 3927, "Metallic Powders Excluding Powders for Hardmetals--Determination of Compactibility

Typically, a cylindrical or rectangular test piece is made by pressing powder in a die, with pressure applied simultaneously from top and bottom.

The pressure required to achieve a specified density is a measure of compressibility.

Compressibility can also be specified as the density achievable at a given pressure. By plotting the density obtained by a series of increasing levels of pressure against these pressures, a compressibility curve is developed.



Compaction of Powder Metals

Two shapes of test pieces have been standardized.

- One is a cylinder having
 - a diameter of 25 mm (1 in.) and
 - a height of 12.7 to 25 mm (0.5 to 1 in.).

- The other is a rectangular bar
 - 12.7 mm (0.5 in.) wide
 - 31.8 mm (1.25 in.) long and
 - 5 to 7 mm (0.2 to 0.3 in.) high.

COMPRESSION RATIO

The compression ratio applies to uniaxial die pressing. It is defined as the ratio of the beight of loose powder to the height of the compact. For a constant cross-section compact, the compression ratio C_R expresses the volume change or density change with a standardized compaction pressure, say $400 \, \text{MPa}_{c}$, and can be calculated from the density ratio, beight ratio, or volume ratio:

$$C_R = \frac{H}{H_O} = \frac{V_L}{V_C} = \frac{\rho_G}{\rho_A}$$

 C_R = compression ratio, dimensionless

H = compacted powder height, m (convenient units: mm)

 H_O = loose-powder fill height, m (convenient units: mm)

 V_C = compacted-powder volume, m³ (convenient units: mm³)

 $V_L = \text{loose-powder fill volume, m}^3 \text{ (convenient units: mm}^3\text{)}$

 ρ_A = loose-powder apparent density, kg/m³ (convenient units: g/cm³)

 ρ_G = pressed green density, kg/m³ (convenient units: g/cm³).



Figure 18-3 (Left) Typical press for the compacting of metal powders. A removable die set (right) allows the machine to be producing parts with one die set while another is being fitted to produce a second product. (Courtesy of Alfa Laval, Inc., Warminster, PA.)



Compaction of Powders

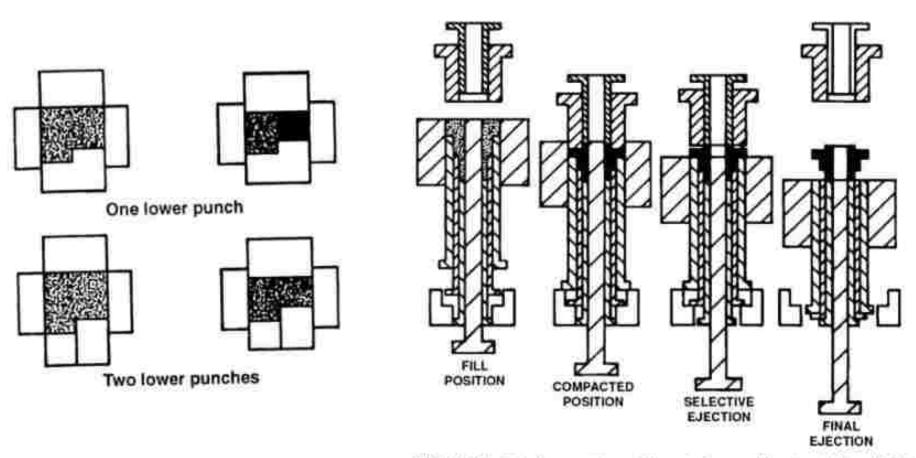
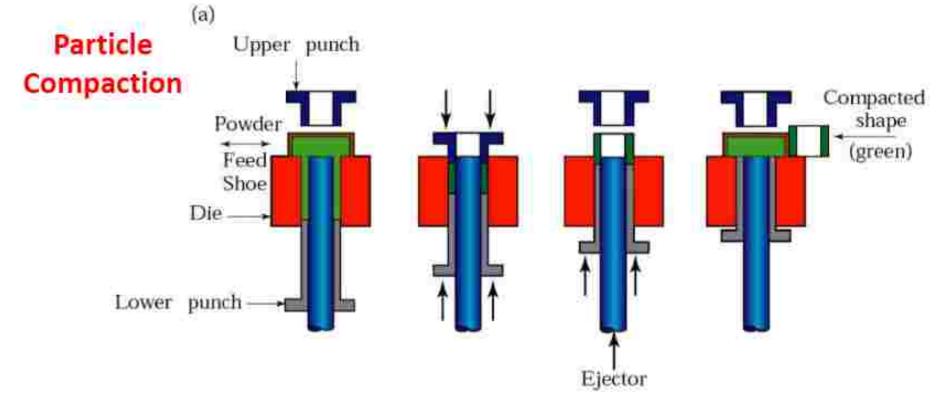
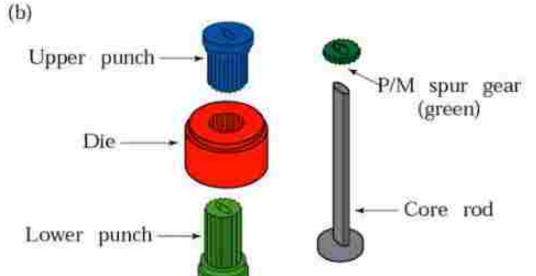


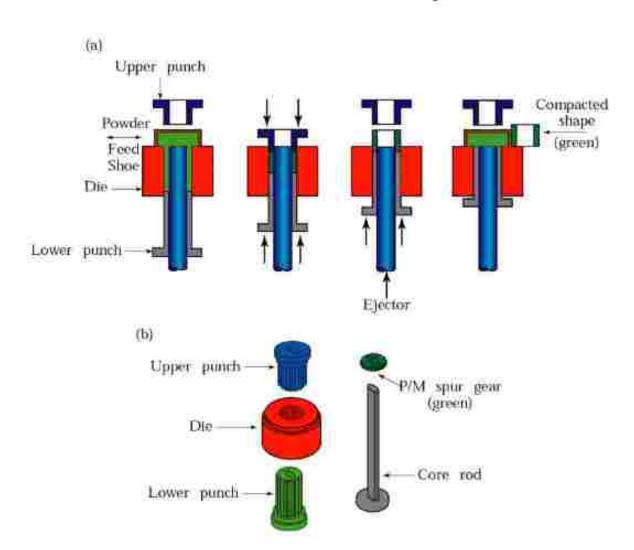
FIG. 11.13 Typical compaction tool elements for a multilevel part. (From Ref. 9.)



(a) Compaction of metal powder to form a bushing. The pressed powder part is called green compact. (b) Typical tool and die set for compacting a spur gear. Source: Metal Powder Industries Federation.



Compaction



- (a) Compaction of metal powder to form a bushing. The pressed powder part is called green compact.
- (b) Typical tool and die set for compacting a spur gear.

Density = (Theoretical density of metal)
$$x \left(1 - \frac{\%porosity}{100}\right)$$

Conductivity = (Theoretical conductivity of metal)
$$\times \left(1-2\frac{\%porosity}{100}\right)$$

Pressure will rearrange and deform contacting particles, inducing an incipient neck between them. The size of that initial neck depends on both the material properties and applied pressure. If no pressure is applied, there is still a small degree of elastic deformation, leading to a small insipient neck. The compaction-induced deformation at the particle contacts produces a circular contact that expands in size with higher pressures. At high relative pressures compared to the material strength, assuming plasticity, the deformed particles will consist entirely of prismatic faces. Prior to formation of prismatic grains, the size of the contact faces can be approximated by a circle of diameter X. The fractional green density f_G and contact size are related as follows: $X = D \left[1 - \left(\frac{f_G}{f_A} \right)^{2/3} \right]^{1/2}$

where D is the particle diameter, and f_A is the fractional apparent density corresponding to X = 0. In uniaxial die compaction, the applied pressure decays with depth below the punch. Accordingly, the compaction-induced initial neck size will vary with position in the green body.

 $D = \text{particle size}, \text{ m (convenient units: } \mu\text{m})$

X = diameter of contact between pressed powders, m (convenient units: μ m)

 f_A = apparent density, dimensionless fraction [0, 1]

 f_G = green density, dimensionless fraction [0, 1].

GREEN DENSITY DEPENDENCE ON COMPACTION PRESSURE (Jones 1960)

In powder compaction the average green density depends on the average compaction stress. Since wall friction decreases the average stress, thicker compacts in the pressing direction will naturally have lower densities. On close scrutiny, there are density gradients within the compact. If it is complicated in shape, especially if there are multiple thicknesses, then the green density can be highly variable within the body. As an approximation, the average fractional green density f_G will depend on the compaction pressure P approximately as follows:

$$\frac{df_G}{dP} = -\Theta \varepsilon = -\Theta(1 - f_G)$$

where ε is the fractional porosity ($\varepsilon = 1 - f_G$), and Θ is a constant that varies with the powder. Rearranging and integrating gives,

$$\ln\left(\frac{1 - f_G}{1 - f_D}\right) = -\Theta P$$

where f_D is the fractional density at the onset of deformation, which is often near the tap density. This equation does not include particle rearrangement, so the addition of a term to include early-stage effects gives,

$$\ln\left(\frac{1-f_G}{1-f_D}\right) = B - \Theta P$$

where B is added to account for particle rearrangement. Modified expressions build from this with terms for deformation and particle hardening. Accordingly, a generic model linking green density to compaction pressure results,

$$f_G = 1 - (1 - f_D) \exp(B - \Theta P)$$

where f_D can be approximated by the apparent or tap density. In some cases, a simplified version can be used to link fractional green density f_G to compaction pressure P,

$$f_G = f_A - A \exp(-KP)$$

where f_A is the apparent density of the powder, and A and K are constants that change with each powder.

A = material constant, dimensionless

B = rearrangement constant, dimensionless

 f_A = fractional apparent density, dimensionless

K = constant, 1/Pa

P = compaction pressure, Pa (convenient units: MPa)

 f_D = fractional density at the onset of deformation, dimensionless

 f_G = fractional green density, dimensionless

 Θ = powder-dependent constant, 1/Pa

 $\varepsilon = 1 - f_G$ = fractional porosity, dimensionless.

GREEN DENSITY DEPENDENCE ON PUNCH TRAVEL

The green density ρ_G in compaction depends on the apparent powder density ρ_A , initial powder fill height H_0 , and final compacted height H as follows:

$$\rho_G = \rho_A \frac{H_0}{H}$$

The compacted height can be expressed as a function of the height change ΔH from the initial height, which is the change in spacing between the upper and lower punches,

$$H = H_0 - \Delta H$$

giving the pressed density as a simple function of the change in punch spacing,

$$\rho_G = \frac{\rho_A H_0}{H_0 - \Delta H}$$

H = final compact height, m (convenient units: mm)

 H_0 = initial powder fill height, m (convenient units: mm)

 ΔH = height change from the initial height, m (convenient units: mm)

 ρ_A = apparent density, kg/m³ (convenient units: g/cm³)

 ρ_G = green density, kg/m³ (convenient units: g/cm³).

FORMING

HOT COMPACTION

ISOSTATIC, EXTRUSION, DIE COMPACTING, SPRAYING, SINTERING

COLD COMPACTION

DIE COMPACTING, ISOSTATIC. ROLLING, INJECTION MOULDING, SLIP CASTING

SINTERING

VACUUM OR ATMOSPHERE

OPTIONAL MANUFACTURING STEPS

RESINTERING, FORGING, COINING, Metal Infiltration OIL IMPREGNATION

Additional Considerations During

- When the pressure is applied by only one punch, the maximum density occurs right below the punch surface and decreases away from the punch
- For complex shapes, multiple punches should be used

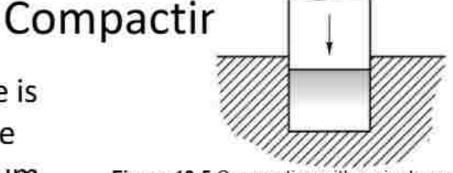


Figure 18-5 Compaction with a single moving punch, showing the resultant nonuniform density (shaded), highest where particle movement is the greatest.

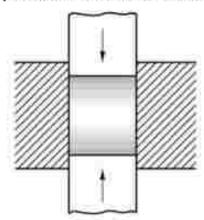
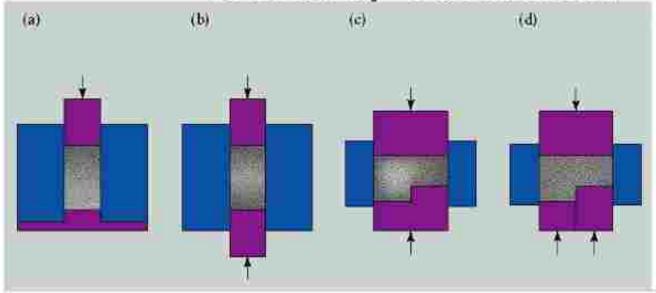


Figure 18-6 Density distribution obtained with a doubleacting press and two moving punches. Note the increased uniformity compared to Figure 18-5. Thicker parts can be effectively compacted.

Density Variation



Density variation in compacting metal powders in different dies:

- (a) and (c) single-action press
- (b) and (d) double-action press.

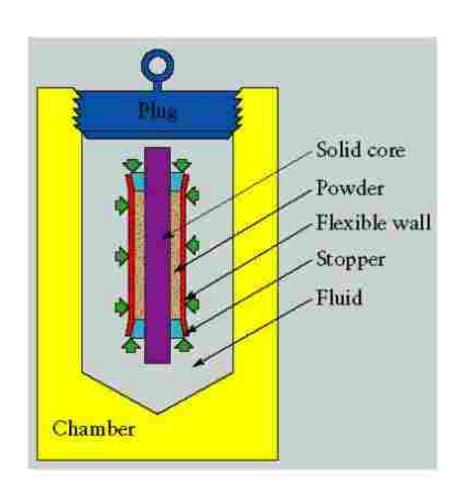
Note in (d) the greater uniformity of density in pressing with two punches with separate movements as compared with (c).

Generally, uniformity of density is preferred, although there are situations in which density variation, and hence variation of properties, within a apart may be desirable.

Complex Compacting

- If an extremely complex shape is desired, the powder may be encapsulated in a flexible mold, which is then immersed in a pressurized gas or liquid
 - Process is known as isostatic compaction
- In warm compaction, the powder is heated prior to pressing
- The amount of lubricant can be increased in the powder to reduce friction
- Because particles tend to be abrasive, tool wear is a concern in powder forming

Cold Isostatic Pressing(CIP)

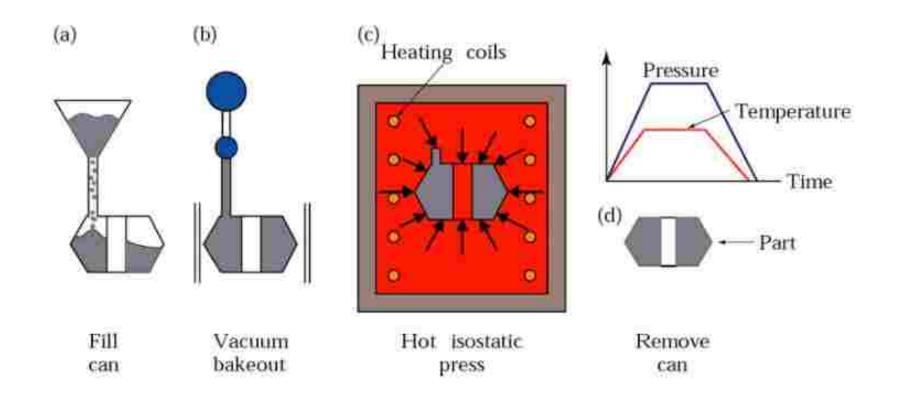


 Schematic illustration of cold isostatic pressing as applied to formation of a tube. The powder is enclosed in a flexible container around a solid core rod. Pressure is applied isostatically to the assembly inside a highpressure chamber.

Hot-Isostatic Pressing

- Hot-isostatic pressing (HIP) combines powder compaction and sintering into a single operation
 - Gas-pressure squeezing at high temperatures
- Heated powders may need to be protected from harmful environments
- Products emerge at full density with unifrom, isotropic properties
- Near-net shapes are possible

Hot Isostatic Pressing(HIP)



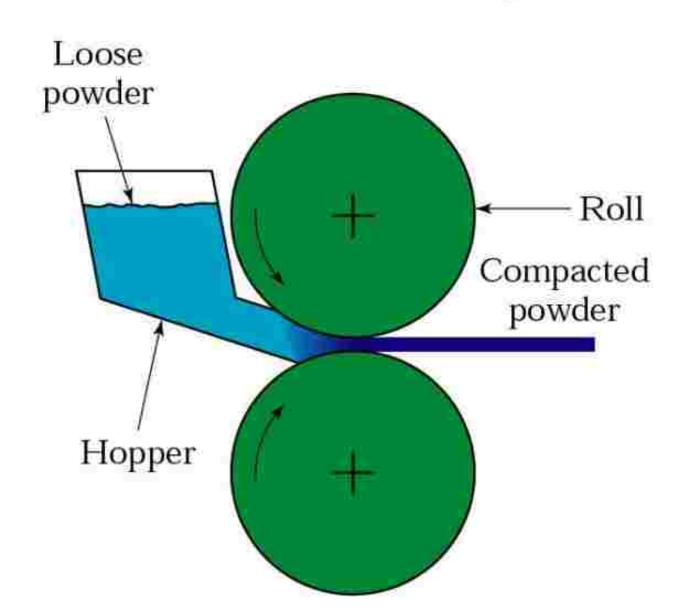
Other compacting and shaping operations

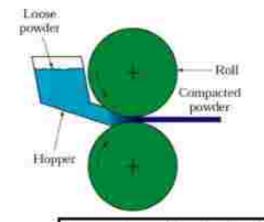
Rolling

Extrusion

Spray Deposition

Powder Rolling





نورد پودر به فشرده کردن پیوسته پودر فلزات به وسیله دستگاه نورد اشاره دارد.

در این پروسه، پودر فلزات از طریق قیف تغذیه میشوند تا ورق ها یا نوارهای خام (زینتر نشده) به روش پیوسته فشرده شوند.

بر روی مواد حاصل پروسه های بیشتری همچون زینتر کردن و نورد دوباره جهت ایجاد محصول نهایی انجام می شود.

می توان موادی با دانسیته بالا و یا با هر سایزی از تخلخل های دلخواه را تولید کرد.

نوارهای تولید شده با نورد کاری پودرها متعاقباً باعث تولید مواد تمیز و کاملاً یکنواخت با اندازه ذرات کوچک میشود.

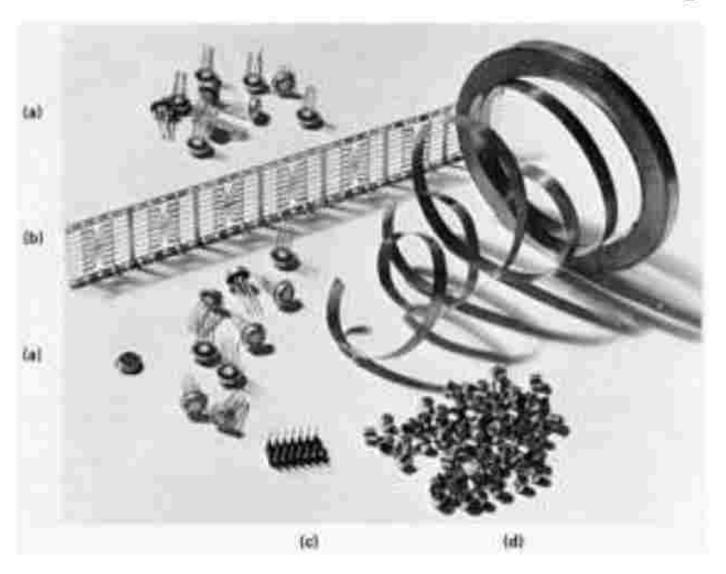
Reference:

Metals Hand book ,Powder Metallurgy, Vol.7, 9th edition, ASM, Metals Park OH, 1984.

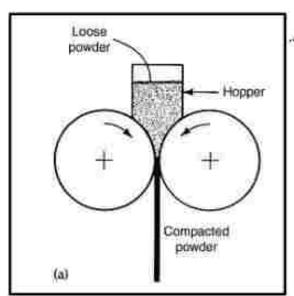
کاربرد مواد نواری فشرده شده با نورد شامل:

- قطعات نواری نیکلی با خلوص بالا برای باطری های تنظیم کننده ضربان قلب
 - نوارهای آهن-نیکل یا انسال سایه
 - قاب های هدایت کننده آهن- نیکل
 - قوطی های Ni- Fe- Co
 - ۵ نوارهای کامپوزیتی Al- Sic
 - نویل های Ti- Al برای تولید کامپوزیت
 - نواری Ni- Ti برای کاربردهای حافظه داری

نمونه هایی از کاربرد نوارهای نیکل فشرده شده



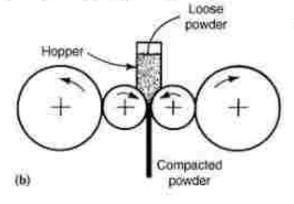
روند توليد: ١.وضعيت غلتک ها

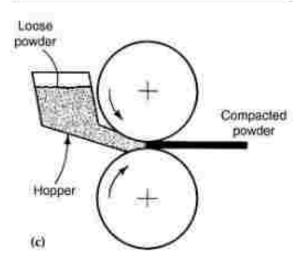


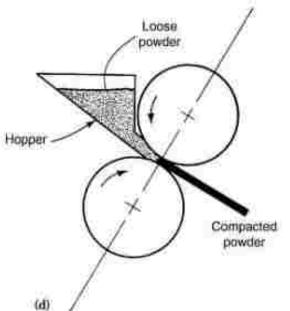
نوع و موقعیت غلتک ها با نوع سیستم انتخاب شده برای تولید نوار مورد نیاز تغییر میکند

غلتک ها می توانند به صورت:

- عمودی (همانند نوارهای سنتی)
 - افقی و یا
- با شیب زاویه دار قرار گیرند.







انتخاب موقعیت غلتک ها با فاکتورهای گوناگون تعیین میشود:

- خصوصیات مواد،
- مقدار نواری که بایستی تولید شود.
 - محصول نهایی مورد نظر،
 - ساختار مخصوص مورد نیاز.

Reference:

Metals Hand book ,Powder Metallurgy, Vol.7, 9th edition, ASM, Metals Park OH, 1984.

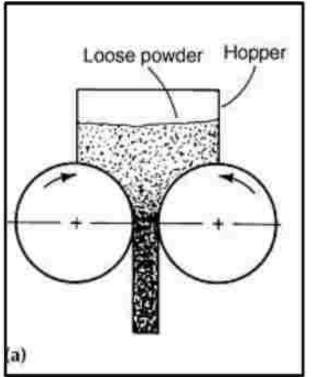
۲.تغذیه پودر

قدم ابتدایی در پروسه فشرده سازی با نورد، تغذیه پودر به درون غلتک هاست.

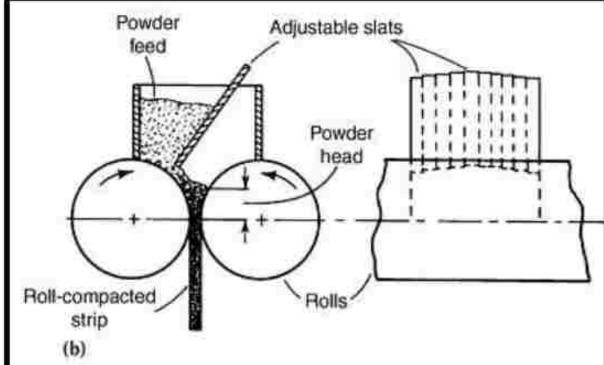
نوار خام بایستی دانسیته یکنواختی را از یک لبه تا لبه دیگر از خود نشان دهد.

دو روش تغذیه به غلتک نوردها

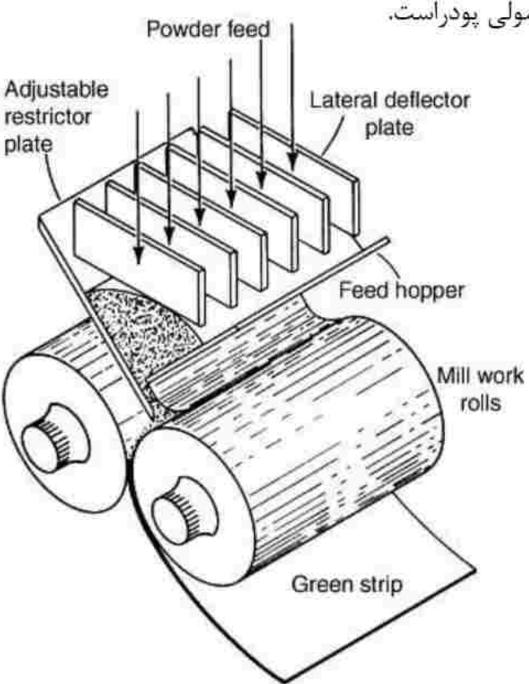
a تغذيه اشباع



b تغذيه غيراشباع



این شکل بیانگر یک تغذیه کننده معمولی پودراست.

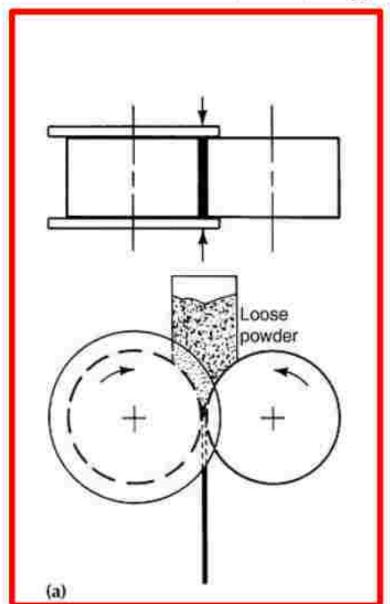


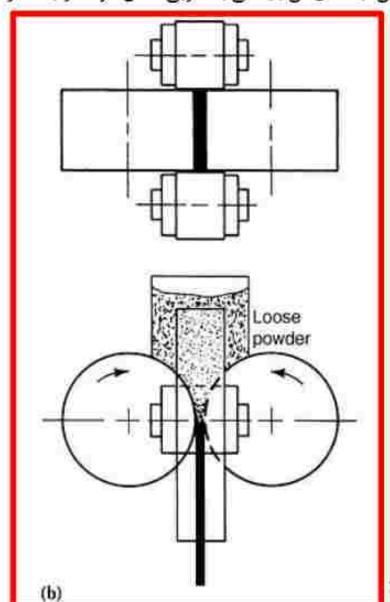
در این نوع طراحی پودر به صورت هموار در عرض قیف پخش می شود.

روشهای حفظ پودر در درون شکاف غلتک ها

٣.کنترل لبه

نوار خام نورد شده از پودر بایستی ضخامت و دانسیته یکنواخت در عرض را دارا باشد. همچنین لبه های آن بایستی به خوبی شکل گرفته و به اندازه وسط نوار دانسیته داشته باشد.

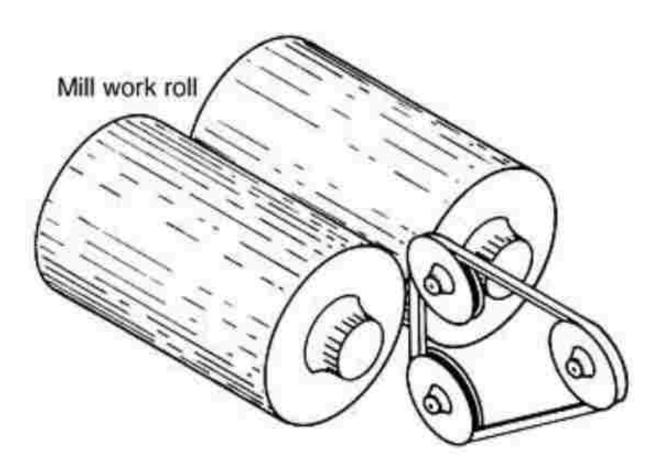




ادامه - روشهای حفظ پودر در درون شکاف غلتک ها

تسمه پیوسته که شکاف ها را در لبه های غلتک می پوشاند نیز روش مؤثری برای جلوگیری از در رفتن پودر است .

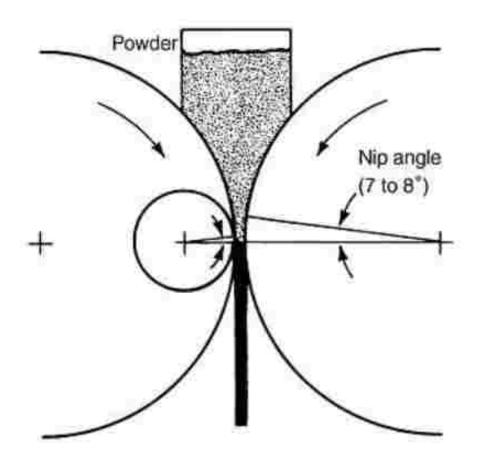
سیستم های دیگری نیز وجود دارند که نوارهایی با لبه های محکم ایجاد می کنند که نیازمند کمی پلیسه برداری در لبه ها خواهد بود.

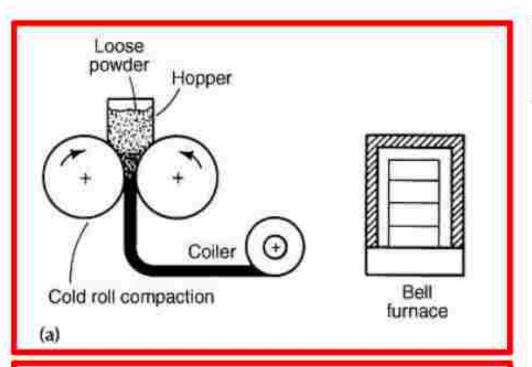


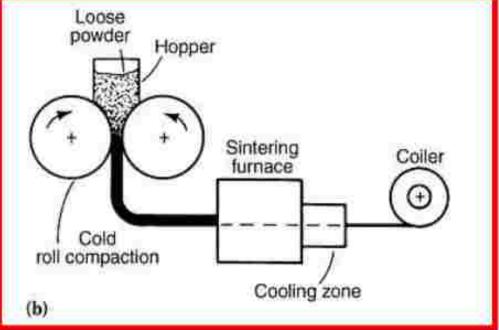
۴.قطر غلتک

در نورد پودر، ضخامت نوار عمدتاً توسط قطر غلتک مورد استفاده تعیین میشود. قطر غلتک از ۹۲۰-۷/۱۲ mm تغییر می کند .

زاویه گاز گرفتگی مورد استفاده د ر مورد نورد پودر همچون زاویه در نورد فلز معمولی می باشد در حدود .توجه داشته باشید که قطر غلتک های بزرگ قوس بزرگتری در روی سطح غلتک به وسیله زاویه گاز گرفتگی ایجاد میشود.







دو روش جابجائی نوار – زمانیکه فشرده سازی با نورد سرد انجام گرفته است.

۵.عملیات تکمیلی

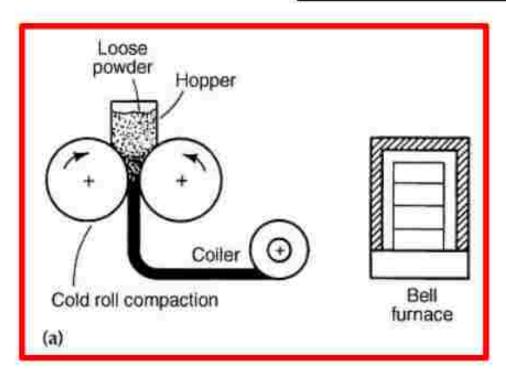
نوار خام یکنواخت تولید شده به وسیله غلتک فشرده سازی، بایستی قبل از دوباره فشرده سازی بیشتر زینتر شود.

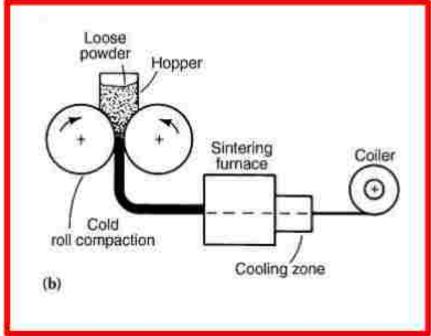
> دانسیته خام معمولاً مابین٪۹۰ ۷۵ دانسیته تئوری است .

۵.عملیات تکمیلی

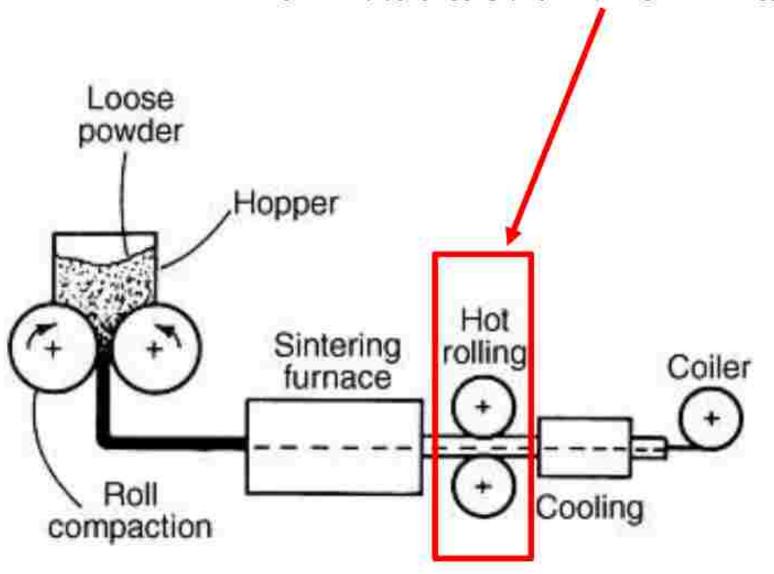
نوار خام یکنواخت تولید شده به وسیله غلتک فشرده سازی بایستی قبل از دوباره فشرده سازی بیشتر زینتر شود. دانسیته خام معمولاً مابین/۹۰ ۷۵ دانسیته تئوری است.

دو روش جابجائی نوار زمانیکه فشرده سازی با نورد سرد انجام گرفته است.

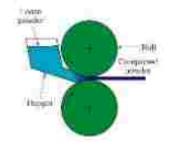




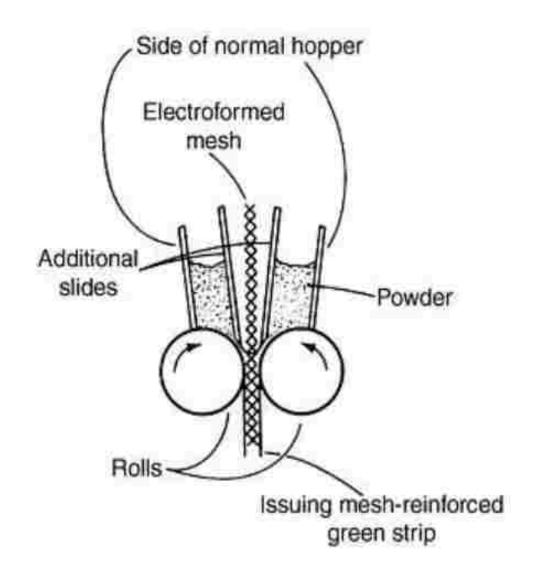
در شکل زیر نوار حاصله از نورد، از میان کوره زینتر عبور می کند در حالیکه هنوز گرم است نورد گرم برای حصول دانسیته کامل انجام شده و سپس نوار سرد و پیچیده میشود.



تولید تجاری



از سال ۱۹۶۸–۱۹۵۸ پروسه نورد پودر توجه قابل ملاحظه ای را در بریتانیا ، آمریکا ، کانادا ، شوروی و ژاپن جلب کرد و یک مرکزی بنا شد تا پروسه های کم قیمت را جهت تولید پودر فلزاتی همچون Al, Fe, Cu, Ni با پروسه های نورد پودر ترکیب کند و روشهای تولید ورق های نازک و لوله ها با شرایط اقتصادی تر گشترش یابند. شکل زیر نشانگر روشی برای تولید نوارهای متخلخل از نیکل خالص برای باطری های قلیائی و سوخت سلولهای الکترونی است. در این روش، مش فرم داده شده در قیف پودر جاگذاری می شود تا به درون نوار نورد شده جا داده شود و الکترود متخلخل بالایی با استحکام خام کافی تولید شود.





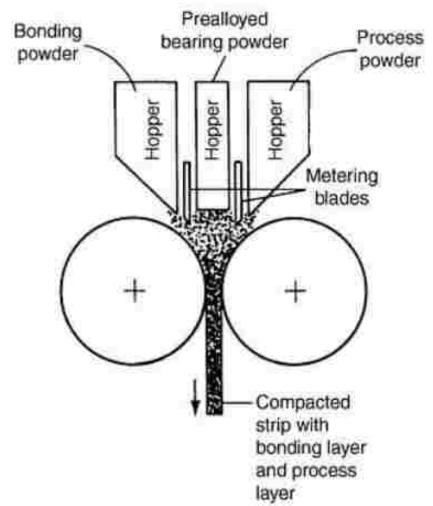


ياتاقان كامپوزيتي

- فشردن با نورد همچنین می تواند برای تولید کامپوزیت استفاده شود.
- مثالی از این مواد کامپیوزیت فشرده شده با نورد، نوارهای بای متال که در تولید خود
 و یا میله انتقال باتاقان استفاده میشود، میباشد.
 - تغییرات در استاندارد وسایل مربوط به اتومبیل منجر به جایگزین شدن این مواد
 جدید به جای ترکیبات روپوش یاطاقان مس سرب سنتی شد.
 - نوار نورد شده شامل لایه ای از Al-8.5Pb-4Si- 1.5Sn-1Cu از پودرهای از قبل
 آلیاژ شده است که بر روی لایه آلومینیوم خالص نورد میشود .

تولید فشرده سازی با نورد برای ایجاد کامپوزیت درشکل زیر آمده است.

- سه تا قیف پودر و تیغه های کنترل کننده جریان پودر مورد نیاز است.
 - این تیغه ها جریان پودر را به درون شکاف نورد کنترل می کنند.
 - سیم پیچ نوار کامپوزیت ایجاد شده سپس زینتر میشود.



ذوب پودربرای تولید یاتاقان کامپیوزیتی

یک کوره دوب القالی توسط آلومینیوم و میلی و سیلیکون عنصری شارز میشود. در دمای بالاتر از دمای متوسط، سرب و قلع به درون مذاب افزوده میشود. دمای کوره تا دمای تک فازی بالا برده میشود کمی بیشتر گرم میشود تا حد اطمینان تأمین شود، جریان هدایشی کوره موجب عمل هم زنی میشود که باعث ایجاد اطمینان از خل شدن قلع و سرب میشود تا یک محلول تک فازی واقعی ایجاد شود .

اتميزه كردن

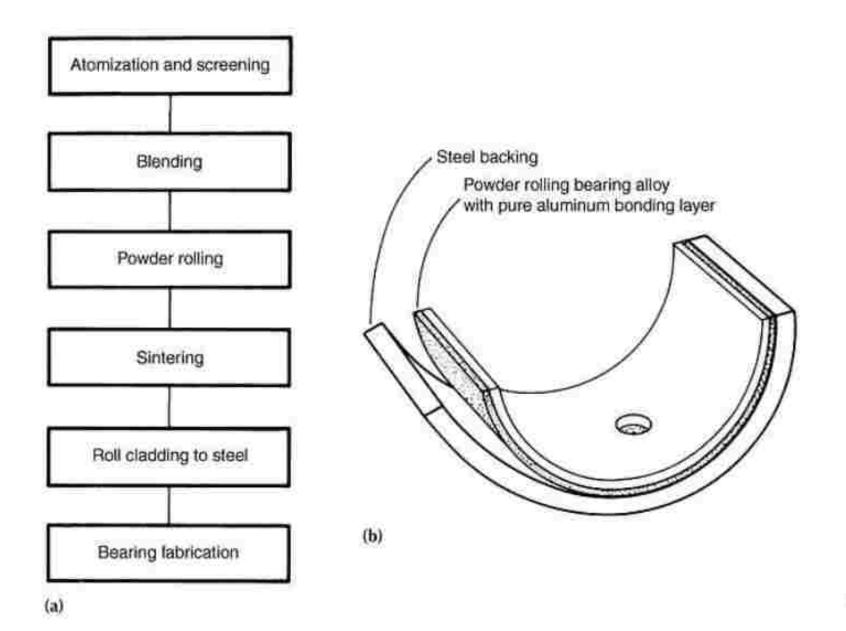
جربان نازک فلز به صورت عمودی به درون محفظه انمیزاسیون ریخته میشود که در آنجا قرآت مجزا شکسته شده و سریعاً متحمد میشود. قرآت بودر به درون جمع کننده ای با ۶متر ارتفاع و ۱۰ متر قطر جمع میشود و سپس آنها از میان یک جدا کننده عبور می کنند تا درات ریز و درشت از هم جدا شده و درات خارج از محدوده را خارج کنند ،

تركيب

روش ترکیبی برای تولید نوار از پیش آلیاری شده آلومیشوم و آلیار سرب انتخاب شده است که در بهایت به بشت فولاد کم کرین، نوار را تورد می کنند.

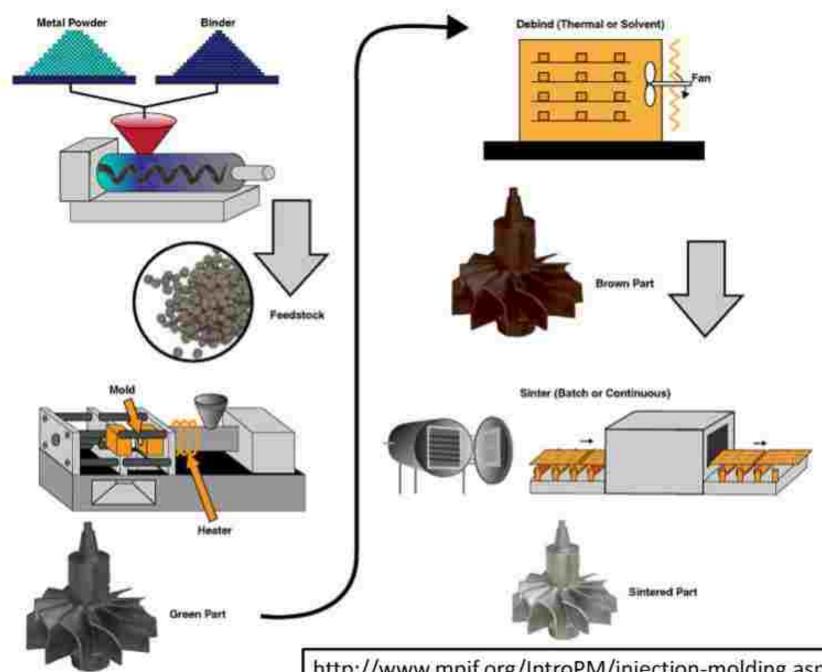
- تجزیه قبلی برای باطاقان های با پایه آلومینیوم کار شده ثابت می کند که آنها برای اتصال مستقیم به سطح فولاد غیرمناسب هستند.
- رمانیکه آلیاز آلومینیوم با فازهای نرمی همچون سرب یا فلع مستقیماً به فولاد پیوند
 می خورد، پیوند واسطه لزوماً شامل بی منظمی های میگروسکوپی است واین بی
 منظمی ها در جاهانی که رسوبات قلع و سرب در نماس مستقیم با فولاد هستند
 آنفاق می افتد.
 - این تقص باعث ایجاد تردی در بنطح شده و به عنوان مکان هایی جهت تسریع در پیشروی ترک خستگی شده وباعث جدا شدن بنوندها تخت شرایط کاری میشود.
 - در نتیجه آلیارهای آلومینیوم کار شده شامل مقدار قابل توجهی قاز ارم است که تقریباً همیشه با لایه بیوندی آلومینیوم با نیکل خالص در مقابل فولاد استفاده میشوند.

نمودار تولید یاطاقان های روکش دار با نورد پودر

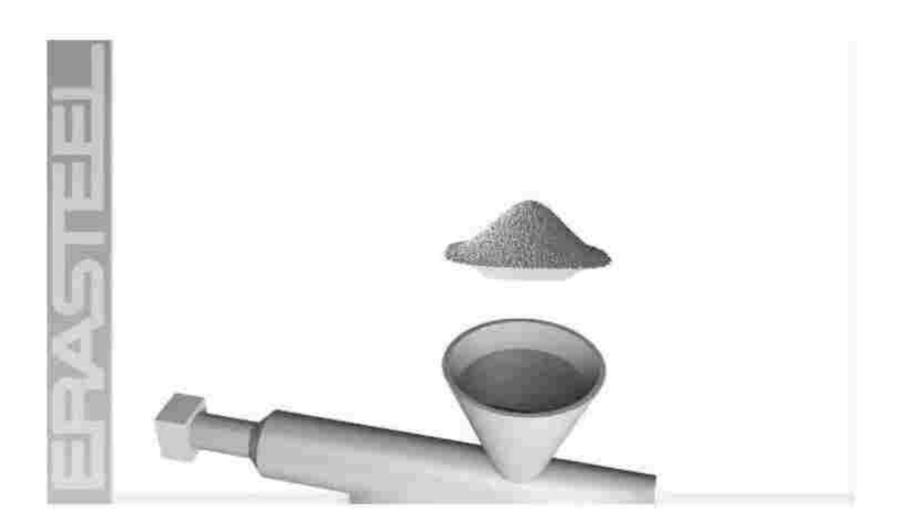


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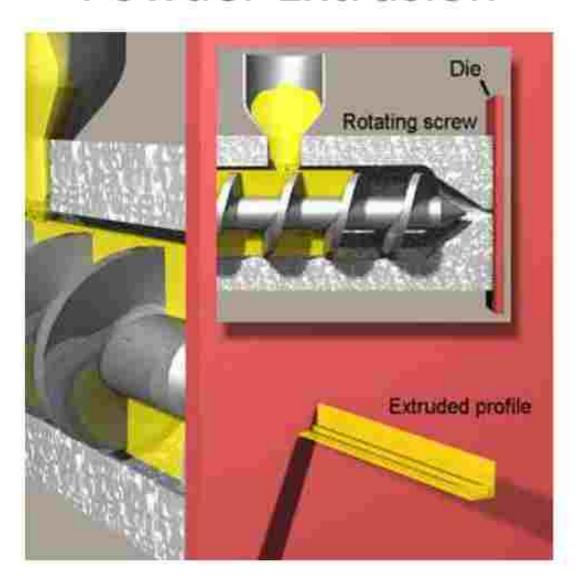
METAL INJECTION MOLDING PROCESS



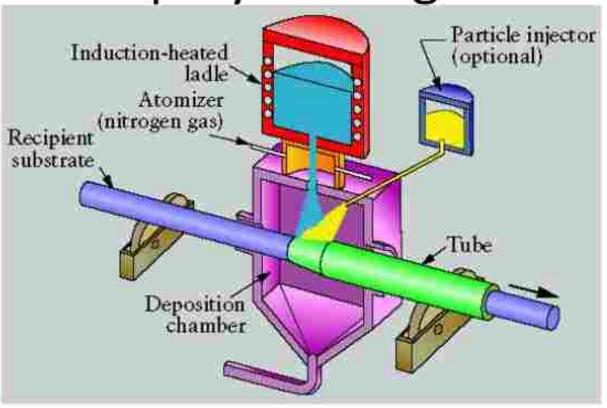
http://www.mpif.org/IntroPM/injection-molding.asp



Powder Extrusion



Spray Casting



Spray casting (Osprey process) in which molten metal is sprayed over a rotating mandrel to produce seamless tubing and pipe..

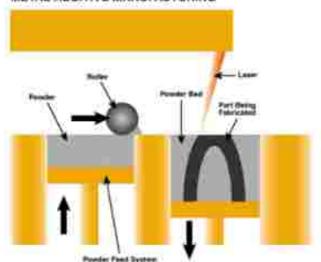
Login or Greate Associat



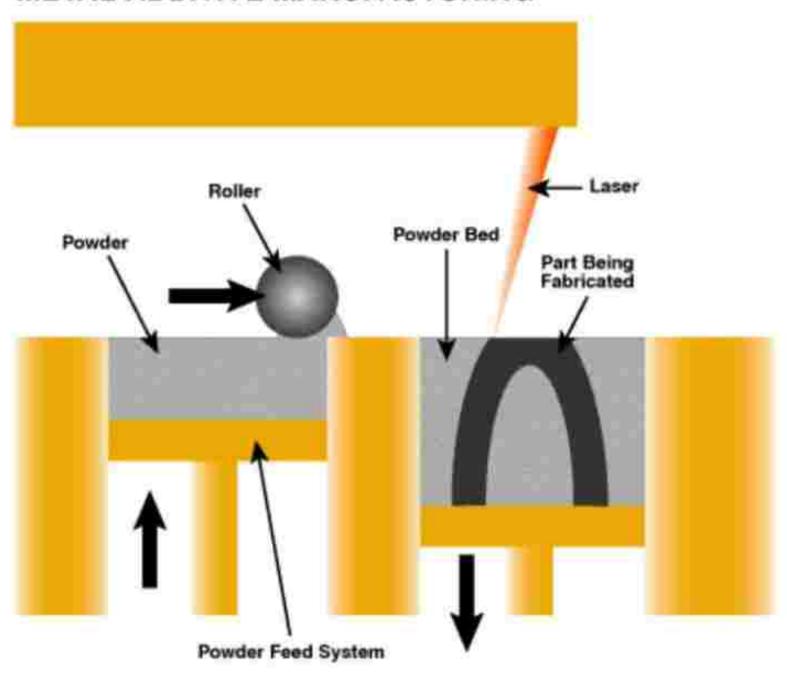
Metal Additive Manufacturing

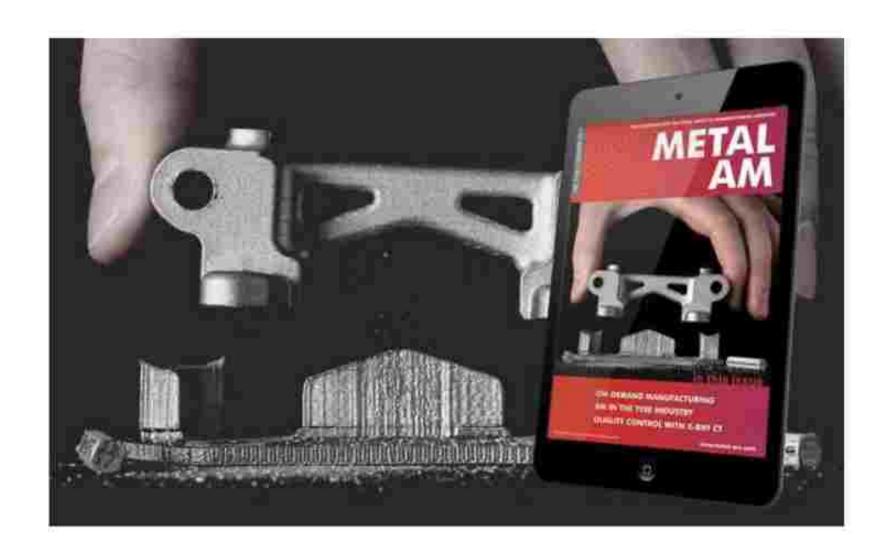
Metal additive manufacturing (MAM), or metal SD pimbles, has the potential to profoundly change the production, time-to-market, and simplicity of components and assemblies. Unlike conventional or subtractive manufacturing processes, such as drilling, which creates a part by removing material, additive manufacturing builds a part using a layer-by-layer process directly from a digital model, without the use of molds or dies that add time, waste material, and expense to the manufacturing process. Additive manufacturing has been used as a design and prototyping tool for decades, but the focus of additive manufacturing is now shifting to the direct production of components, such as medical implants, aircraft engine parts, and jewelry.

METAL ADDITIVE MANUFACTURING



METAL ADDITIVE MANUFACTURING





http://www.metal-am.com/metal-additive-manufacturing-magazine/

FORMING

HOT COMPACTION

ISOSTATIC, EXTRUSION, DIE COMPACTING, SPRAYING, SINTERING

COLD COMPACTION

DIE COMPACTING, ISOSTATIC. ROLLING, INJECTION MOULDING, SLIP CASTING

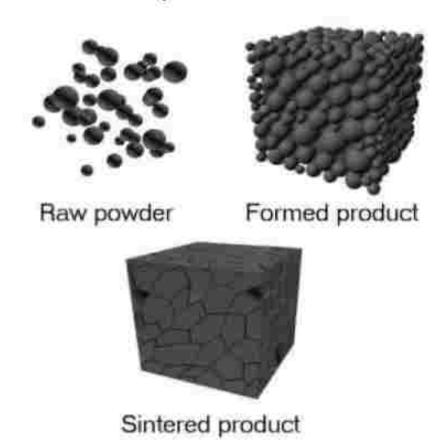
SINTERING

VACUUM OR ATMOSPHERE

OPTIONAL MANUFACTURING STEPS

RESINTERING, FORGING, COINING, Metal Infiltration OIL IMPREGNATION

Powder Compaction and Sintering





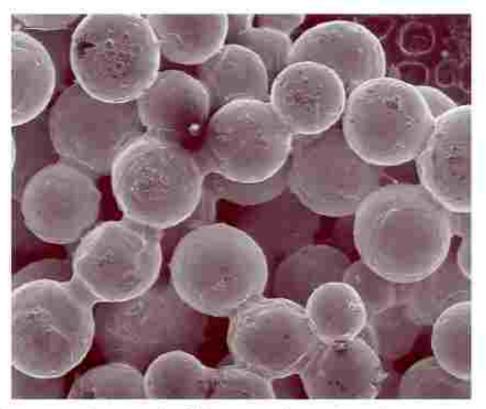


Figure 1.1 Scanning electron micrograph of bronze sphere sintering.



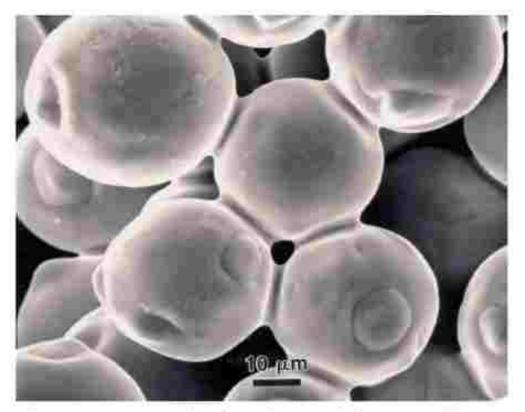
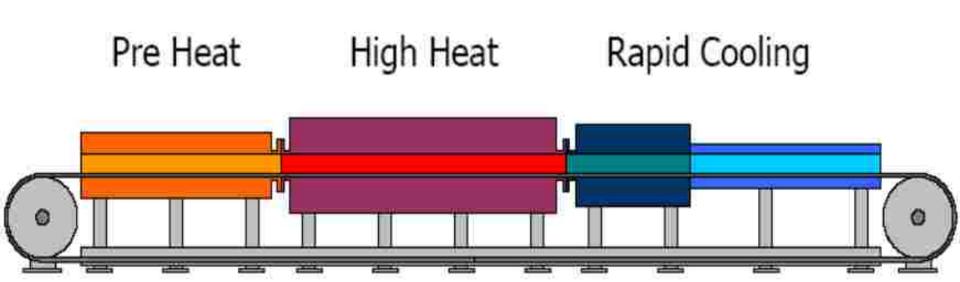


Figure 4.1 Scanning electron micrograph of neck growth between 32 μm nickel spheres during sintering at 1050°C (1323 K) for 30 min in a vacuum. Prior to sintering the particles were loosely packed into a crucible.



Continuous Furnace Temperature Profile

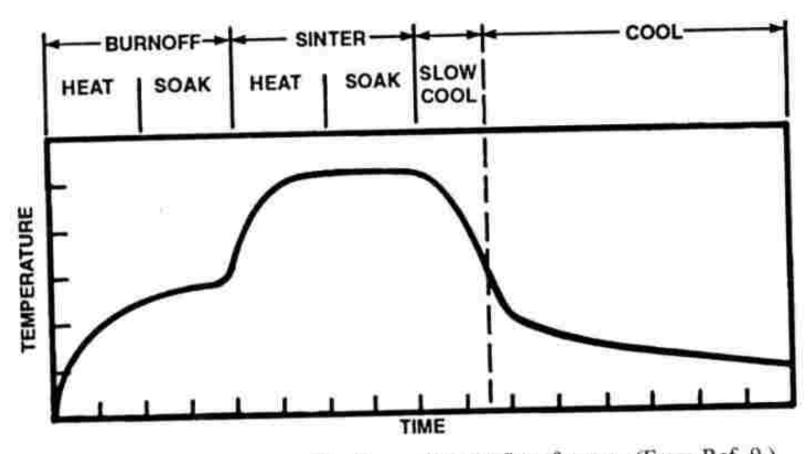
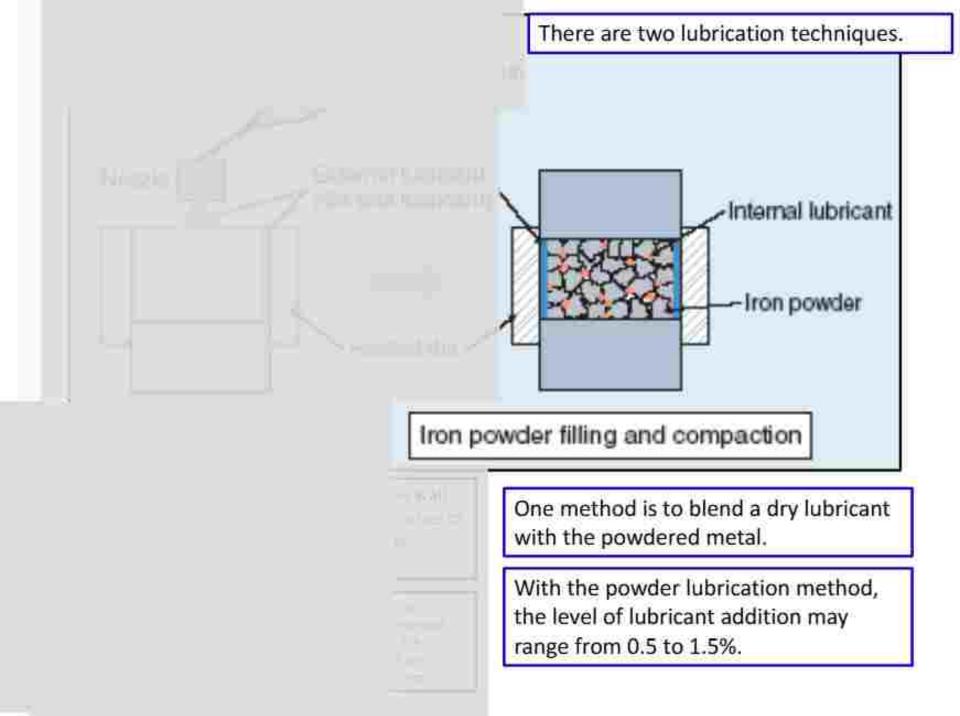


FIG. 11.18 Temperature profile for continuous-flow furnace. (From Ref. 9.)

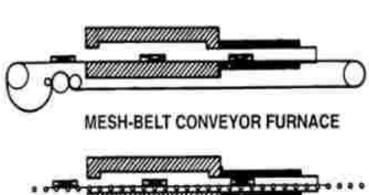


Characteristics of Sintering Furnaces

Design for Powder Metal Processing

TABLE 11.7 Typical Operating Temperatures for Sintering Furnaces

| Furnace type | Maximum operating temperature (°C) | |
|-----------------|------------------------------------|--|
| Continuous flow | 1977/do s | |
| Belt | 1150 | |
| Pusher | 1150 | |
| Roller-hearth | 1150 | |
| Walking beam | 1650 | |
| Batch type | | |
| Bell | 2800 | |
| Elevator | 2800 | |
| Vacuum | 2800 | |







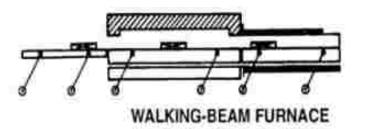
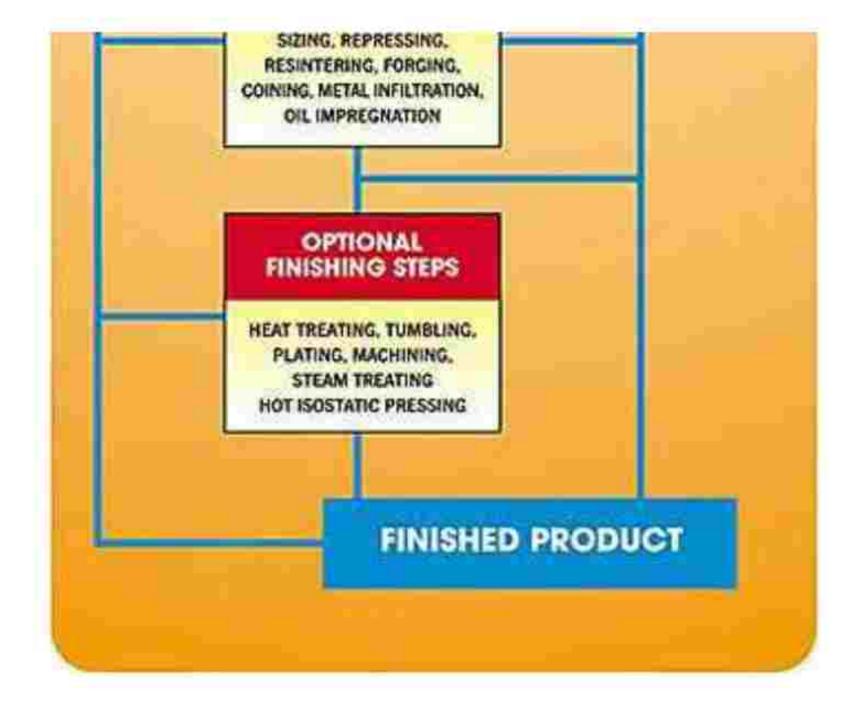


FIG. 11.17 Continuous-flow furnaces. (From Ref. 8.)

Sintering temperature required for different group of materials

Table 6-4: Sintering temperatures used for selected metallic and non-metallic materials (partly according to [6.4])

| Material | | Sintering temperature |
|--|---|-----------------------|
| Aluminium alloys | | 590 620 |
| Copper | | 600 900 |
| Bronzes | | 740780 |
| Brass | | 890 910 |
| Nickel | | 1000 1150 |
| Iron-carbon, iron-copper carbon | | 1120 |
| Iron-copper-nickel carbon | | 1120 |
| Iron-manganese copper | | 1120 |
| Iron-copper-nickel molybdenum (Distaloy) | | 1120 1200 |
| Iron, iron-copper, iron-copper nickel | | 1120 1280 |
| Iron-chromium, iron-chromium copper | | 1200 1280 |
| Iron-manganese | | 1280 |
| Iron chromium carbide (Cr ₂ C ₂) | > | 1280 |
| Iron-vanadium carbide | > | 1280 |
| Iron-tungsten carbide | > | 1280 |
| Iron-manganese-vanadium-molybdenum- carbon | > | 1280 |
| Iron-manganess-chromium-molybdenum- carbon | > | 1280 |
| Electrolytic iron, carbonyl -iron, aluminium-nickel iron (soft and | | |
| hard magnets) | | 1200 1300 |
| Hardmetals | | 1200 1400 |
| Heavy metals (heavy metals of W) | | 1300 1600 |
| Tungsten alloys | | 1400 1500 |
| Nitrides | | 1400 2000 |
| Molybdenum disilicide (MoSi, heating elements) | | to 1700 |
| Aluminium oxide (Al,O, - ceramic cutting tools) | | 1800 - 1900 |
| Tungsten, molybdenum, tantalum | | 2000 2900 |



Secondary operations are performed to increase density, improve accuracy, or accomplish additional shaping of the sintered part

- Repressing pressing the sintered part in a closed die to increase density and improve properties
- Sizing pressing a sintered part to improve dimensional accuracy
- Coining pressworking operation on a sintered part to press details into its surface
- Machining creates geometric features that cannot be achieved by pressing, such as threads, side holes, and other details

Effect of double compacting on mechanical properties of Cr-Mo prealloyed PM steels

Maziyar AZADBEH









AS-REPRESSED PROPERTIES OF DOUBLE COMPACTED Cr-Mo AND Mo PREALLOYED PM STEELS

M. Azadbeh, H. Danninger, Ch. Gierl

Abstract

Obtaining higher densities is a major target in powder metallurgy, and there are many parts that require higher densities than can be obtained by conventional die compaction. One method for solving this problem is double compaction, which seems to be a potentially attractive production route in PM also for Cr-Mo and Mo alloyed steels. The objective of this research was to find the optimum conditions for obtaining higher densities and minimise interconnected porosity. The effect of the temperature of intermediate annealing was studied for Cr-Mo and Mo prealloyed steels, respectively, and the optimum temperature range was identified. It could be observed that the density of repressed samples increases with higher annealing temperature up to a certain limit, softening of the matrix as well as generation of free porosity through some C dissolution being helpful. On the other hand, too high annealing temperatures should be avoided either, since too much dissolution of carbon during annealing is detrimental, resulting in harder and less deformable compacts. Even at this latter condition, however, no repressing cracks were observed.

Keywords: sintered steels, double pressing, density, annealing

STUDY ON THE INTERPARTICLE BONDING OF DOUBLE PRESSED PM ALLOY STEELS

M. Azadbeh, H. Danninger, Ch. Gierl

Abstract

Double pressing is a common technique for improving the relative density of sintered steel components, with resulting beneficial effect on the mechanical properties. To eliminate the work hardening introduced by the first pressing, an intermediate anneal is done before the second pressing, typically at temperatures at which carbon is not yet dissolved. In this work, interparticle bonding caused by the intermediate anneal is described through fractographic techniques for prealloyed steels Fe-Mo-C and Fe-Cr-Mo-C and compared to the fracture surfaces after final sintering. It showed that after annealing, interparticle bonding is significantly more pronounced in the Mo alloyed steel than in the Cr-Mo grade, apparently as a consequence of the more stable surface oxides in the latter material. After final sintering however there are virtually no differences and with regard to mechanical properties the Cr-Mo material offers considerable advantages.

Keywords: sintered steels, double pressing, presintering, interparticle bonding

Investigation of influences of alloying elements and sintering temperature on the properties of high strength low alloyed sintered steel

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Department of Materials Engineering, Sahand University of Technology, Tabriz, Iran, P.O. Box 51335-1996

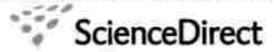
Abstract

Producing parts with high density and improved mechanical properties is one of the most important aims of powder metallurgy process. There are many factors for attaining modified properties in sintered parts but among them controlling type and quantity of alloying elements and manufacturing parameters such as compacting pressure, sintering temperature are the most effective.

In this research, two series of Cr-Mo prealloyed powders with 1.5% and 3% chromium contents were used, and then influences of manufacturing parameters on physical and mechanical properties were investigated. The results show that by compacting pressure and sintering temperature increment and chromium content decrement, density is increased and subsequently physical and mechanical properties of low alloyed sintered steels are improved.

Keywords: Low alloyed sintered steel, sintering temperature, density, electrical-conductivity, mechanical properties.

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Effect of Intermediate Annealing Temperature on Response of Repressing to Densification of Pre-Alloyed Cr-Mo Steel

Amir Kalantari, Maziyar Azadbeh (Department of Materials Engineering, Sahand University of Technology, Tabriz 53317-11111, Iran)

Abstract: Increasing density is one of the important factors for producing high quality powder metallurgy (PM) parts, which has beneficial effect on mechanical properties. One of the common techniques for achieving this goal is double compacting, which seems to be a potentially attractive method in PM route, also for Cr-Mo alloyed-steels. The objective of this research was to investigate the effect of first compacting pressure and intermediate annealing temperature on attaining higher densities and minimum interconnected porosity for Cr-Mo pre-alloyed steel. The effect of mentioned parameters was studied by measuring density, transverse rupture strength and macrohardness of repressed samples. The results show that for each first compacting pressure, the density range of repressed samples increases with the increasing annealing temperature up to a certain limit, due to C dissolution which causes free porosity and further densification. Annealing temperatures higher than optimum one should be avoided, since too much carbon dissolution results in harder and less deformable compacts. On the other hand, with regard to repressed density and other resulted properties, the amount of first compacting pressure offers considerable advantage in obtaining higher level of density and consequently improved mechanical properties.

Key words: double compacting: repressed density: repressed transverse rupture strength; intermediate annealing temperature

Material:



Fe-Cr-Mo-C (Fe - 1.5% Cr - 0.2% Mo)+ 0.6% C



| Apparent density | 2.85 g/cm ³ | |
|------------------|------------------------|--|
| ow | 26 sec/50g | |
| rticle size | 45-150 µm | |
| Chemi | cal analysis, % | |
| C | < 0.01 | |
| Chromium | 1.50 | |
| Molybdenum | 0.20 | |



Fe-Mo-C

(Fe - 1.5% Mo)+ 0.6% C

| | Properties of Fe-Mo powder | | |
|------|----------------------------|-----------------------|--|
| | Apparent density | 3.1 g/cm ⁵ | |
| | Flow | 25 sec/50g | |
| | Particle size | 45-150 µm | |
| 25 5 | Chemical analysis, % | | |
| | C | < 0.01 | |
| 400 | Molybdemim | 1,50 | |

Specimen:

Rectangular bars: 55 mm x 10 mm x 10 mm

1st Compacting at 600 MPa

Compacting



Sintering

1st sintering at different teperatures



Compacting

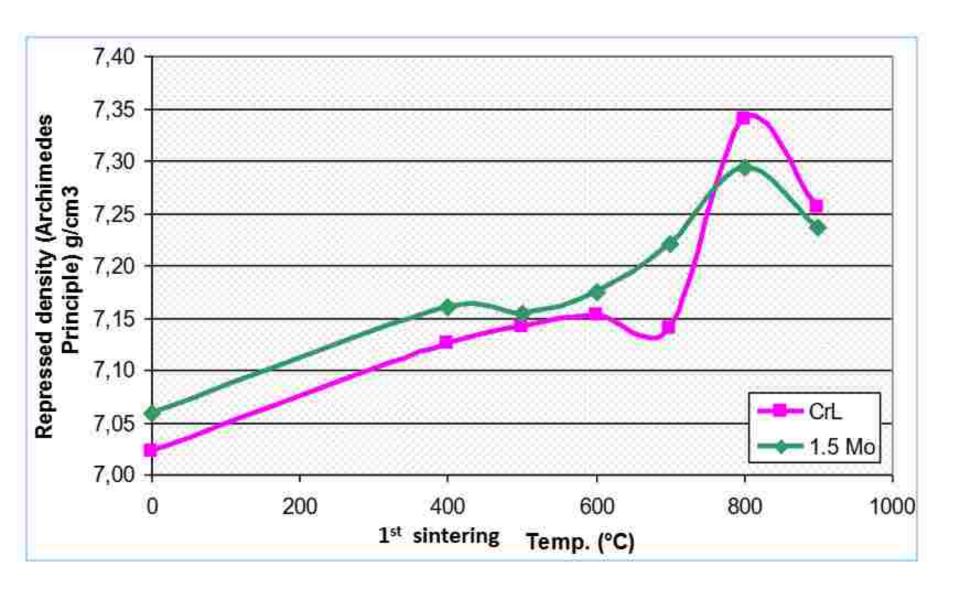
2nd Compacting at 600 MPa

Sintering:

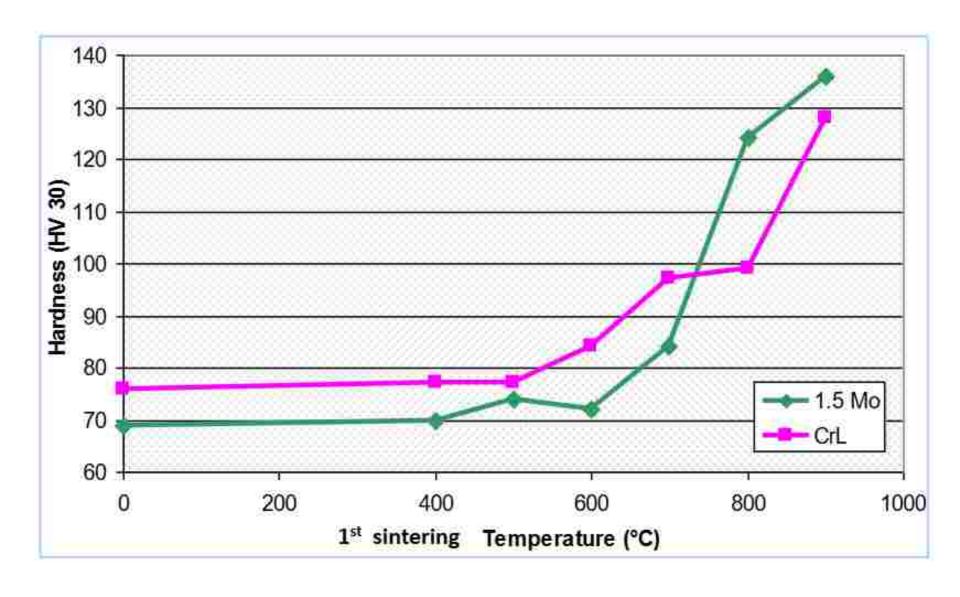
Sintering at 1250°C was carried out in a laboratory furnace with gas tight superalloy retort in **flowing high purity nitrogen** (min. 99.999 purity, flow rate 2 l/min).

Sintering

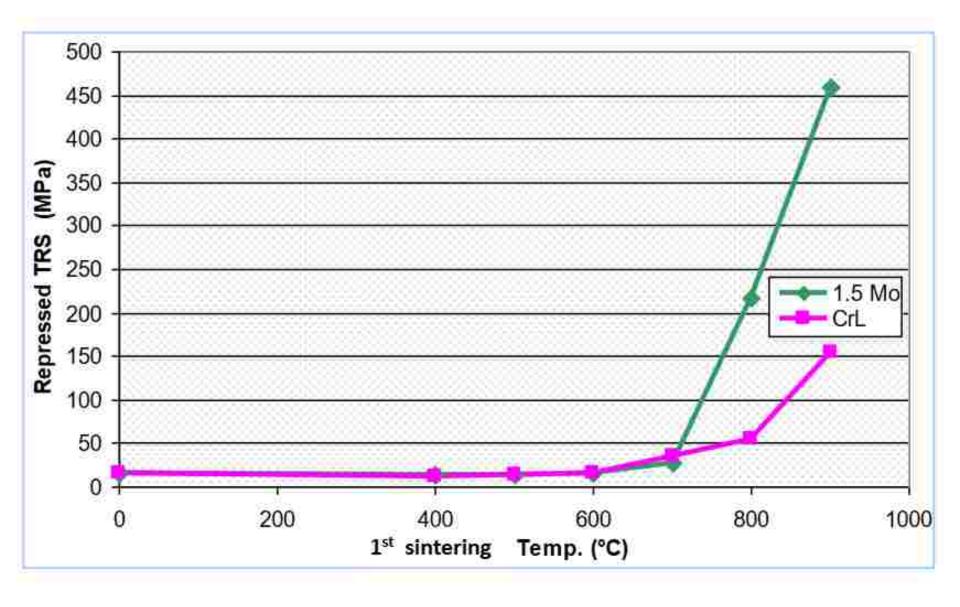
Final sintering at 1250 °C



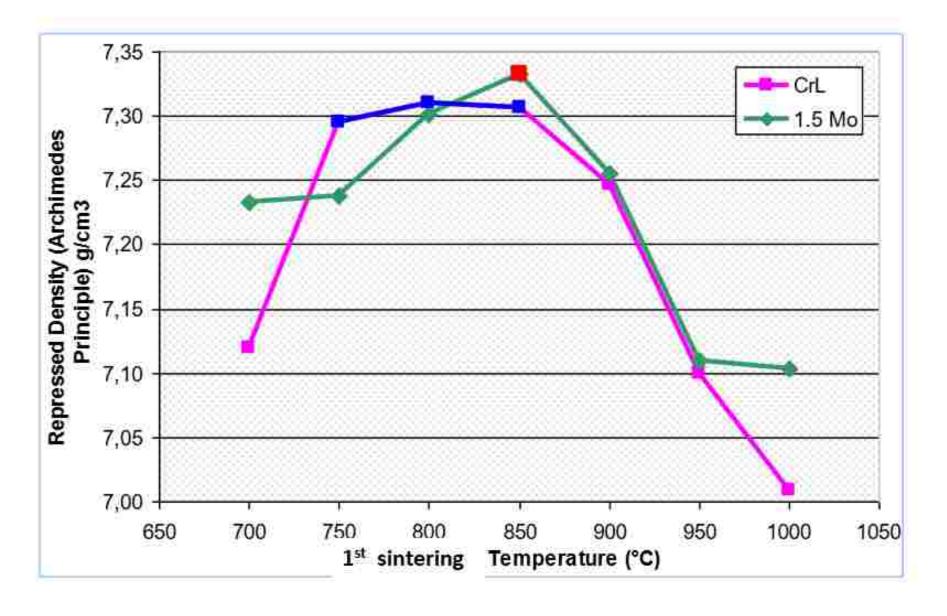
Effect of 1st sintering on repressed density (obtained by Archimedes principle). 600 + 600 MPa, intermediate anneal 30 min in



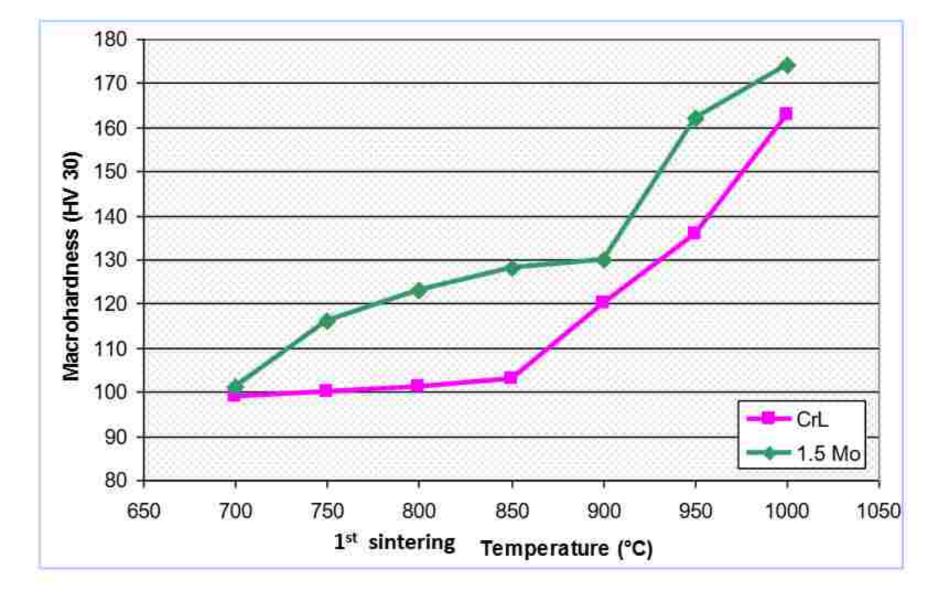
Effect of intermediate sintering on macrohardness of repressed Fe-Cr-Mo-C and Fe-Mo-C.



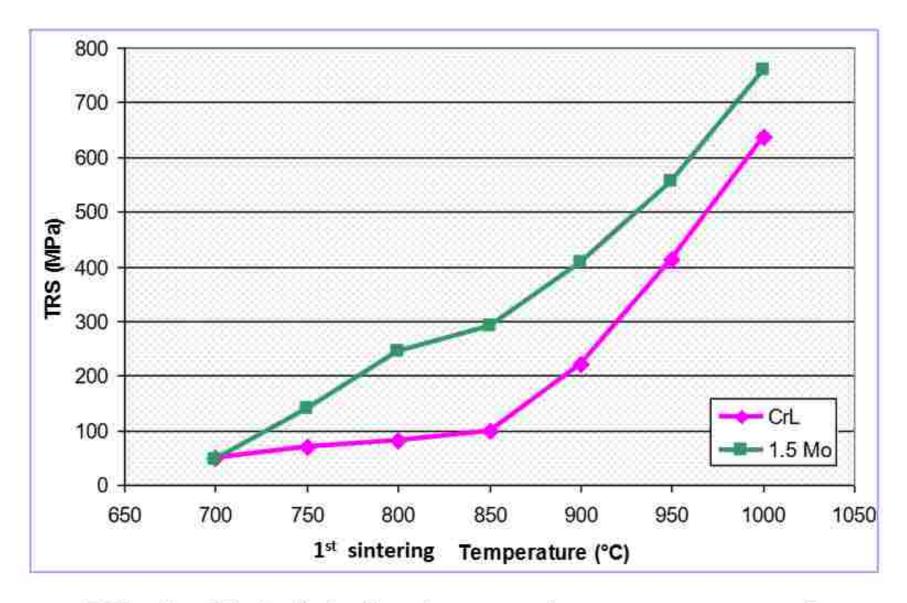
Effect of the intermediate sintering temperature on transverse rupture strength of repressed Fe-Cr-Mo-C and Fe-Mo-C



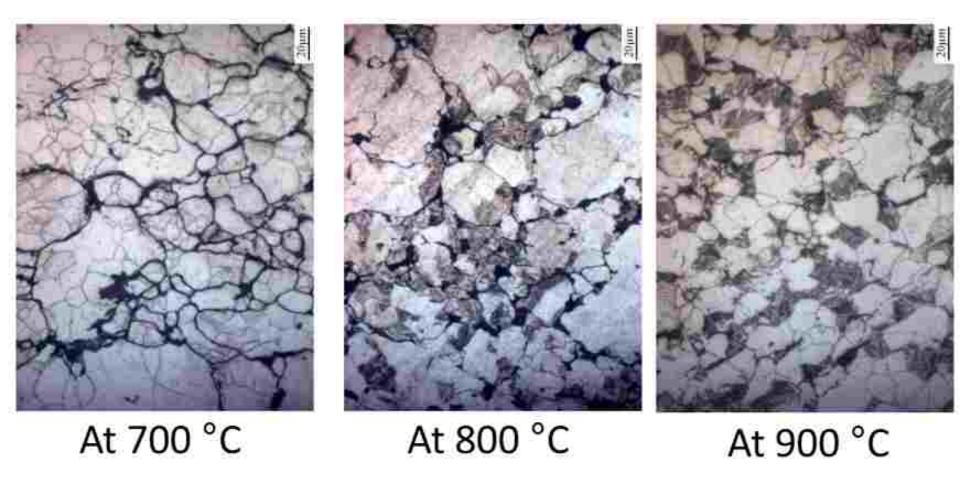
Effect of 1st sintering temperature on repressed density (obtained by Archimedes principle) after repressing at 600MPa.



Effect of 1st sintering temperature on macrohardness of repressed samples

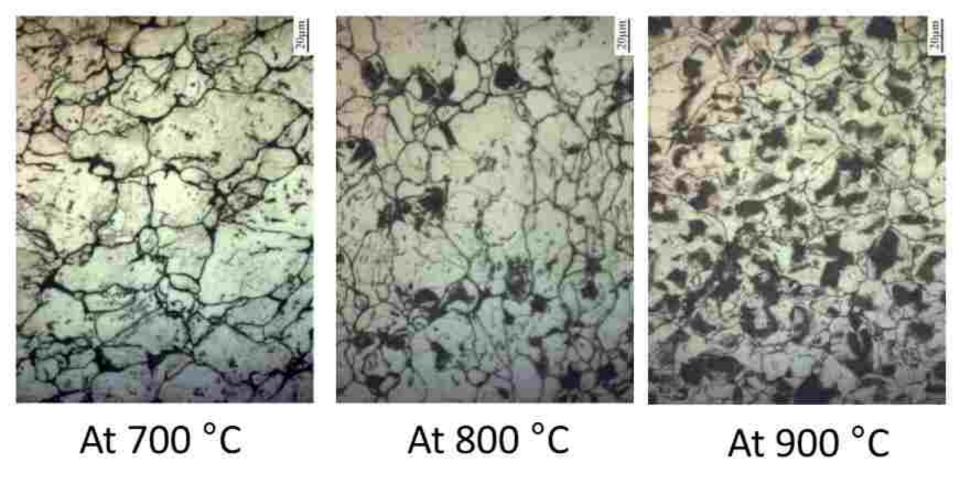


Effect of 1st sintering temperature on repressed transverse rupture strength of samples

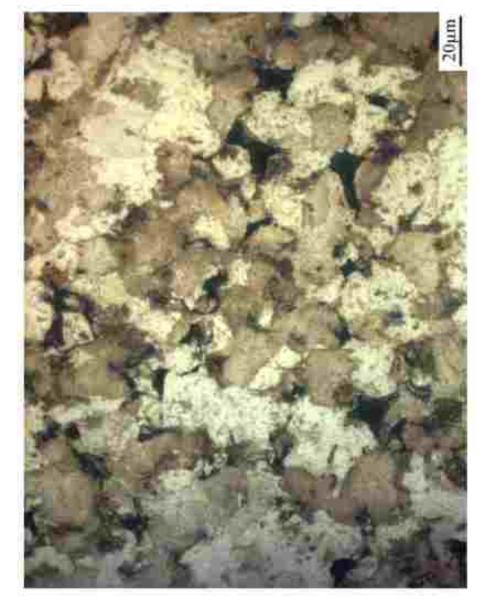


Microstructure of Fe-Mo-C,
compacted at 600 MPa and

1st sintered at the indicated temperature for 30 min, and
repressed at 600 MPa. (As repressed)



Microstructure of Fe-Cr-Mo-C, compacted at 600 MPa and 1st sintered at the indicated temperature for 30 min, and repressed at 600 MPa. (As repressed)



Green - 1st sintered at 1000°C

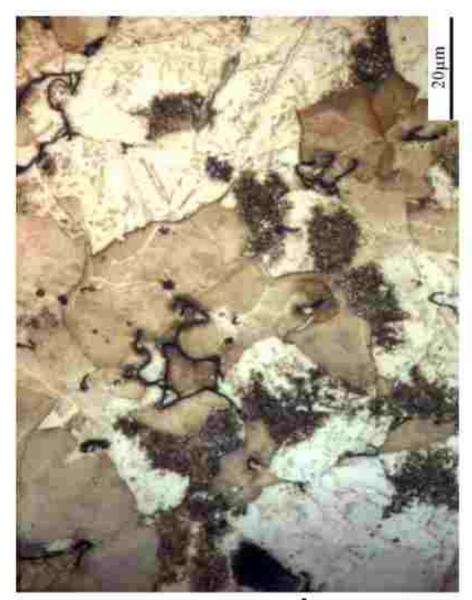


Repressed

Fe-Mo-C



Green - 1st sintered at 1000°C



Repressed

Fe-Mo-C



Green - 1st sintered at 1000°C



Repressed

Fe-Cr-Mo-C

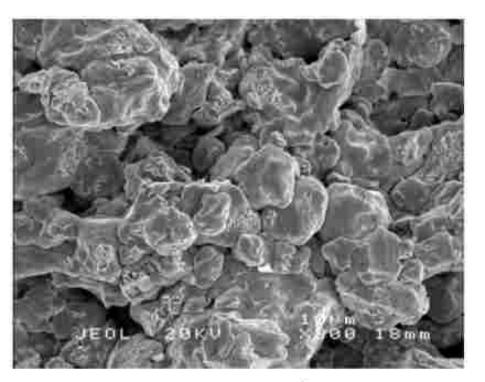


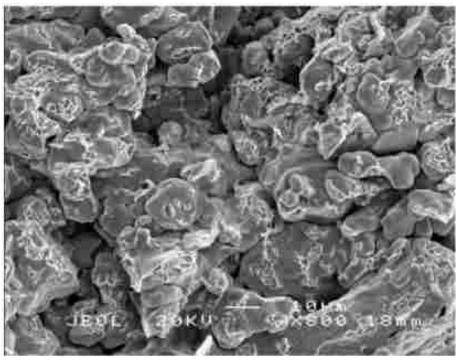
Green - 1st sintered at 1000°C



Repressed

Fe-Cr-Mo-C



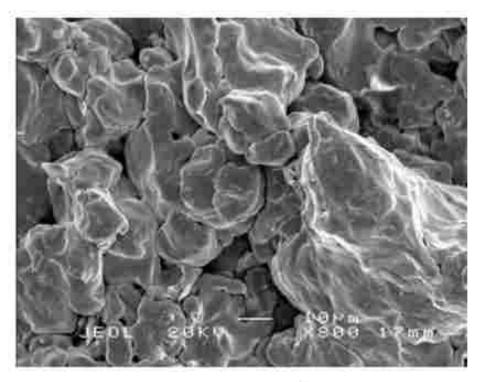


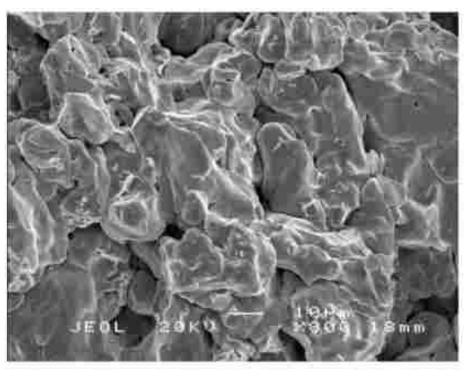
At 800 °C

At 900 °C

Fracture surface of repressed Fe-Mo-C,
compacted at 600 MPa and

1st sintered at the indicated temperature for 30 min, and
repressed at 600 MPa. (As repressed)



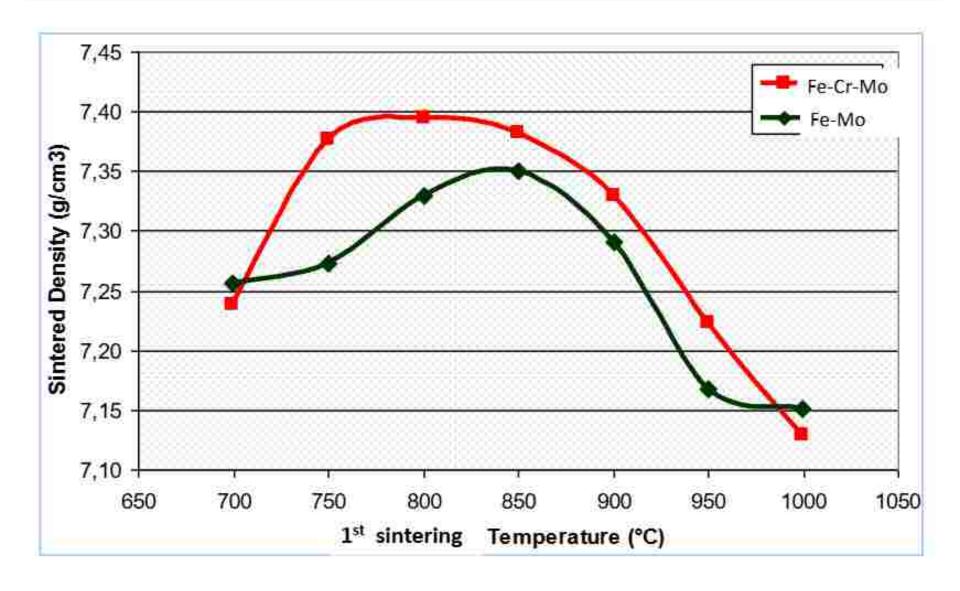


At 800 °C

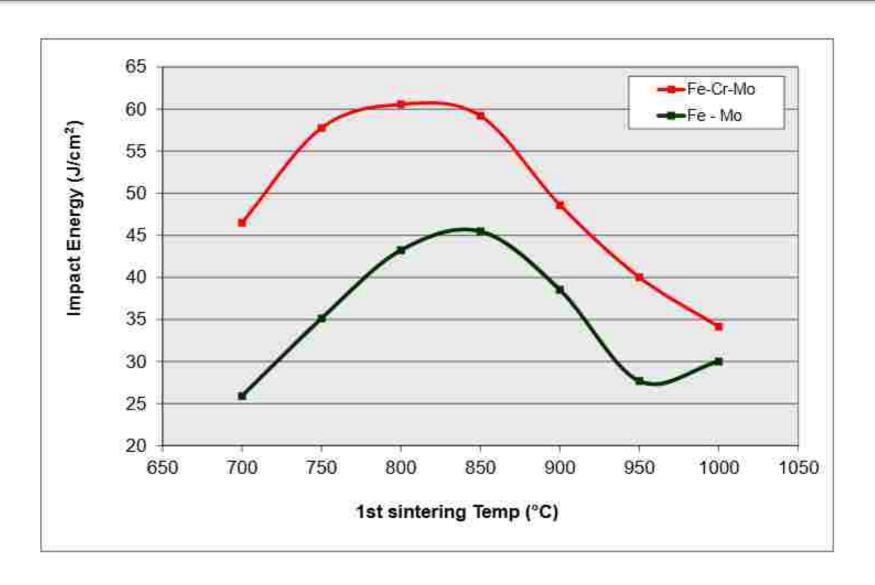
At 900 °C

Fracture surface of repressed Fe-Cr-Mo-C, compacted at 600 MPa and 1st sintered at the indicated temperature for 30 min, and repressed at 600 MPa. (As repressed)

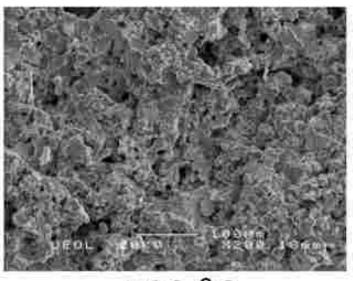
Effect of intermediate sintering temperature on sintered density



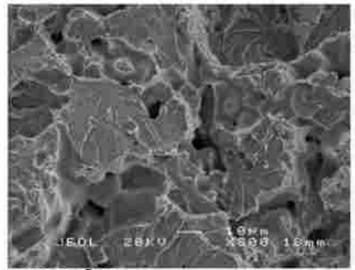
Effect of intermediate sintering temperature on impact energy



Fracture surface of repressed and then sintered Fe-Mo-C as a function of 1st sintering tempareture



500°C



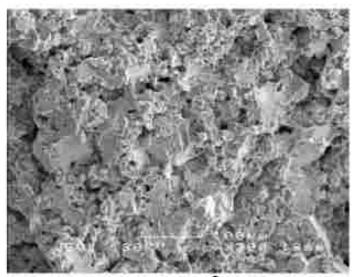
500 °C, higher magnification

Compacted 600 MPa,

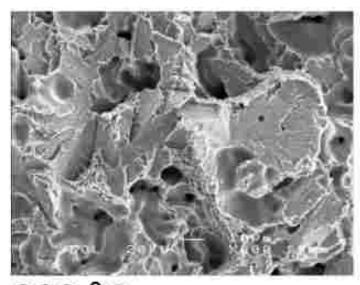
1st sintering -30 min in N₂

Repressed at 600 MPa,

sintered-60 min at 1250°C in N₂

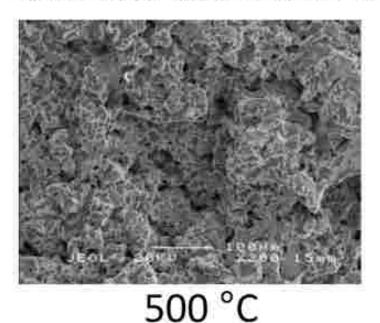


800°C



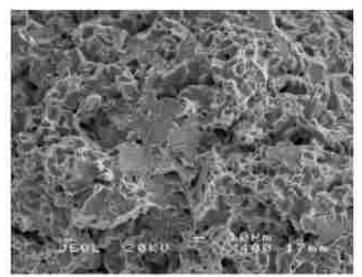
800 °C, higher magnification

Fracture surface of repressed and then sintered Fe-Cr-Mo-C as a function of 1st sintering tempareture



Compacted 600 MPa,

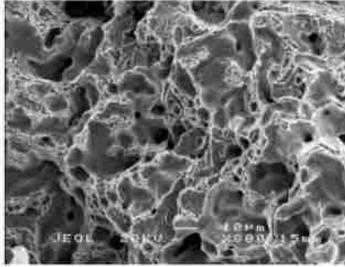
1st sintering -



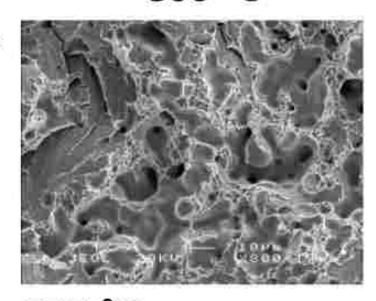
Repressed at 600 MPa,

sintered-60 min at 1250°C in N₂

800°C



500 °C, higher magnification



800 °C, higher magnification

- PM alloys are true engineered materials, and in order to achieve the highest density, hardness and impact energy in PM steels, it is necessary to optimise the properties of both interparticle and intraparticle regions.
 - By proper manufacturing parameters, higher sintered densities such as 7.39 and 7.35 g/cm3, respectively for Fe-Cr-Mo-C and Fe-Mo-C using described method under double compacting at 600 MPa and sintering at 1250°C for 1 h in N₂ are achievable.
 - ☐ This technology offers opportunities for development of high performance PM materials.
- Improved mechanical properties are achieved by such a process, which is a consequence of the increased sintered density.



- ➤ This trend can be improved with increasing annealing temperature to the specific temperature around 800 - 850°C.
 - ☐ At higher examined annealing temperature the repressibility drops.
 - Because the microstructure obtained after annealing at higher temperature is comparable to very low temperature sintered bodies, the macrohardness of green bodies before repressing increases and repressing will be done in harder samples since the repressing has to overcome the already built up sintering necks, that can cause defects and microcracks, which may be not healed during sintering process.
 - On the other hand annealing at higher temperature is uneconomic.



Sizing

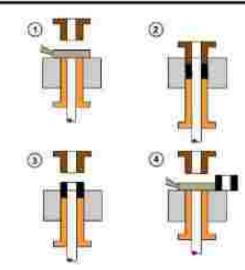
- Secondary operations are performed to increase density, improve accuracy, or accomplish additional shaping of the sintered part
- Repressing pressing the sintered part in a closed die to increase density and improve properties
- Sizing pressing a sintered part to improve dimensional accuracy
- Coiming pressworking operation on a sintered part to press details into its surface
- Machining creates geometric features that cannot be achieved by pressing, such as threads, side holes, and other details

Sizing: بطور نوعی امکانپذیر است که قطعات زینتر شده با دقتی که دارای تلرانس ابعادی برابر

□ 0.0508 mm/mm در جهت عمود بر مسیر فشرده شدن و 0.1016 mm per mm در جهت موازی با مسیر فشرده شدن و شدن تهیه کرد.

که این عمل توسط فشردن مجدد Repressing بعد از زینتر کردن میسر می شود که بنام سایز کردن مشهور است.





Impregnation and Infiltration

- It can be exploited to create special products by filling the available pore space with oils, polymers, or metals
- Impregnation: The term used when oil or other fluid is permeated into the pores of a sintered PM part
 - Common products are oil-impregnated bearings, gears, and similar components
 - An alternative application is when parts are impregnated with polymer resins that seep into the pore spaces in liquid form and then solidify to create a pressure tight part
- Infiltration: An operation in which the pores of the PM part are filled with a molten metal
 - The melting point of the filler metal must be lower
 - Involves heating the filler metal in contact with the sintered component so capillary action draws the filler into the pores
 - The resulting structure is relatively nonporous, and the infiltrated part has a more uniform density, as well as toughness and strength

Impregnation

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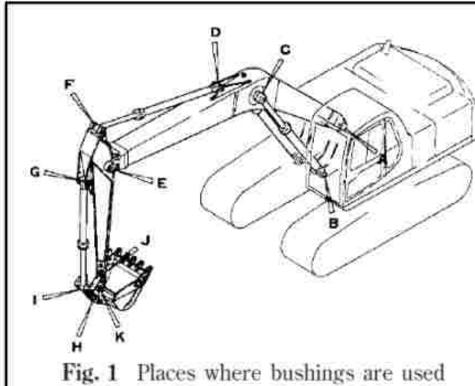






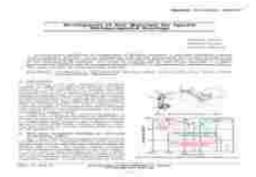


Photo 3 Appearance of BMRC bushings



MINARIE TRECOVERS. REPORT

Development of New Materials for Special Off-Impregnated Bearings



Total



Photo 3 Appearance of BMRC bushings

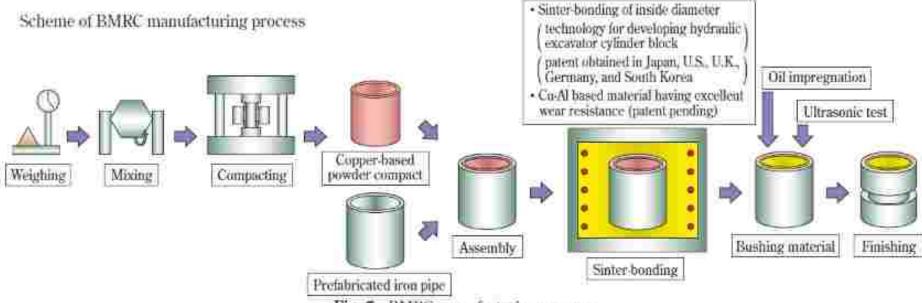


Fig. 5 BMRC manufacturing process

Bimatel Bushing







Steel + CuPbaSn,



Steel + CuPb_{xx}



Steel + AlSn₁₀Cu





Impregnation and Infiltration

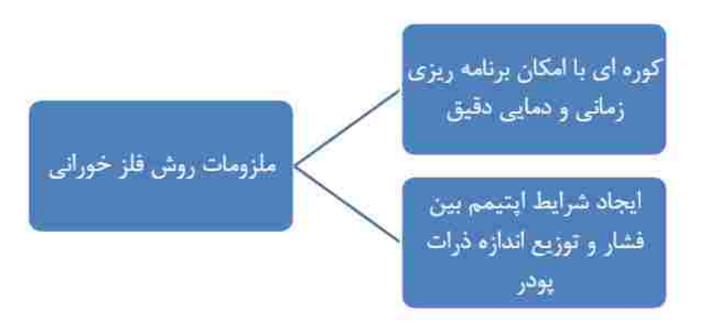
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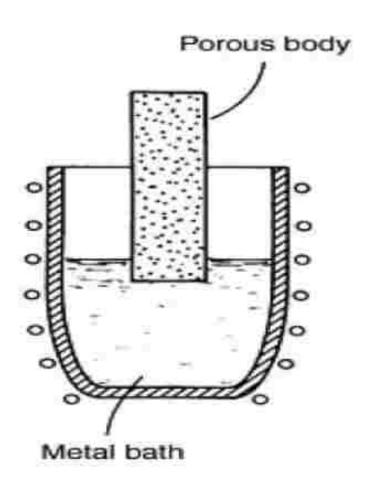






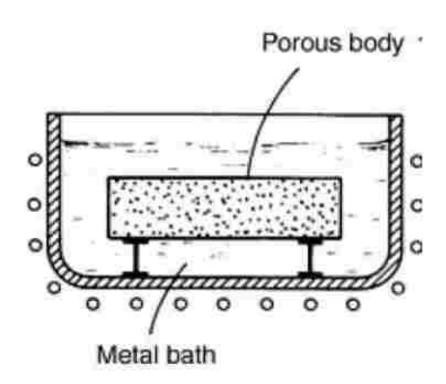
فلزخوراني:۶۰

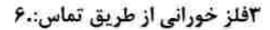


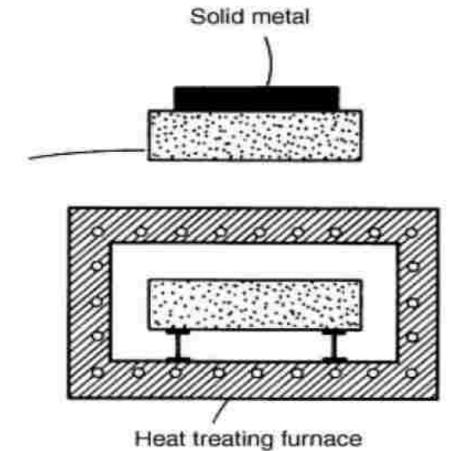


 افلز خورانی از طریق(موئینگی-غوطه ور ۶۰ شدن):

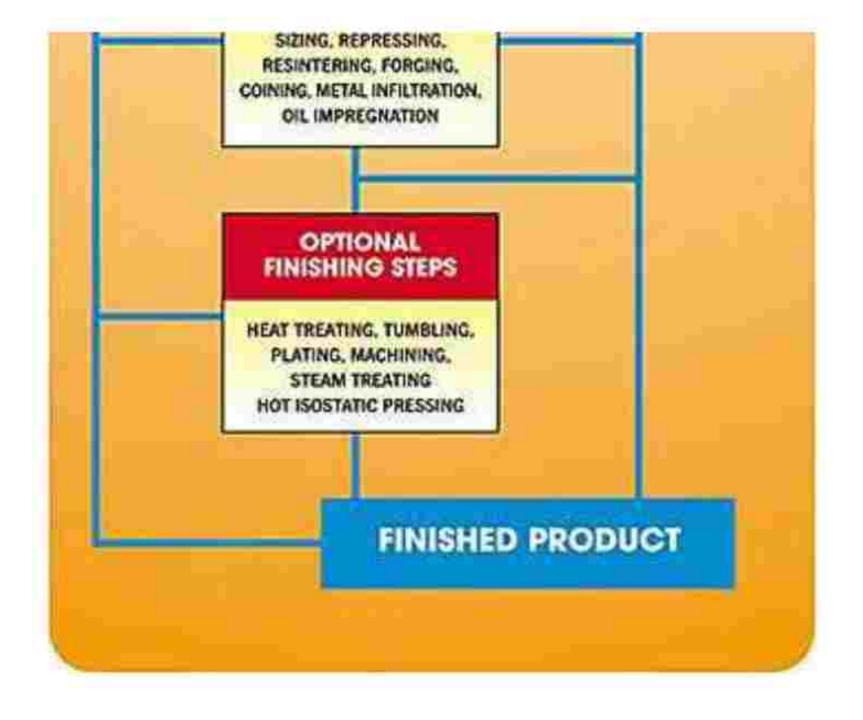
کفلز خورانی از طریق غوطه وری .
 کامل:





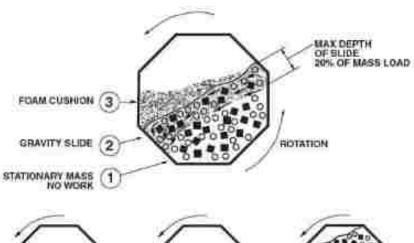






BARREL TUMBLING

PARTS RATIO - 4 MEDIA TO 1 PARTS BY VOLUME. USE AS A STARTING POINT CHANGE UP/DOWN TO ACHEIVE DESIRED RESULTS.

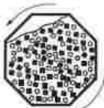




20 - 30' FILL VOLUME PART ON PART TUMBLE, NO GLIDE



LONGEST SLIDE FASTEST FINISH



70 - 80 FILL VOLUME SHORT SLIDE SLOW FWISH BURNISH OF DELICATE PART LOADING

COMPOUND REQUIREMENTS

- 1. THICK DURABLE 2. HEAVY FDAM CUSHION FOR SOIL RETENTION

- 3. HIGH LUBRICITY TO MAINTAIN A GOOD DEPTH OF SLIDE 4. SPEED, VARIABLE FROM 50 TO 200 S.F.M. SLOWER SPEED BURNISHING HIGHER SPEED DEBURRING



Machine Tumbling

By althory our hared tuestiers either for part on-part processing or in continuous with media; subtem can effectively rises and deliver it very with range of parts and part slows. Tampling can also be used as a dearing process to remove stanging oils. Next their scale, and beers, of the inown speculies.

The tenting process is rather simple. Facts are placed to a hartwood tenting living, which with sticking depoints and rest with thirty, and often as strackin multis. The bases are this inbut if of anywhere from 20 to 40 filte for a determined amount of time depending on the decined fluids.

The harrier is designed to lift and snove the parts as that they do not remain to the same formation. Resighest the pricess. The parts width knock represently against our another, and or the residu. This number, using with the right adregateds, realist for an effective decading process. Funding its the laked process for smaller, more rigid parts.

Common Robbins after advanced through toroiding lockable

- w Staffering
- a Democring
- 4 OHERW
- · Restrational.
- forfers berlinning.
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- . Named Big Read Street scools
- De Gashing
- · Brightsoning parts.
- · Property for further finishing routing.

By witting barrel turblers and using within mills or part on part promotes, we hav effectively

these and deliver various standparts. We can remove stamping siles, burn and beat scale all in one sp

the files Transfering or Vitamory Forbitions































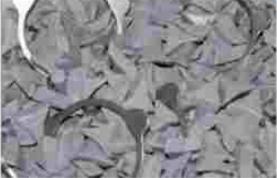




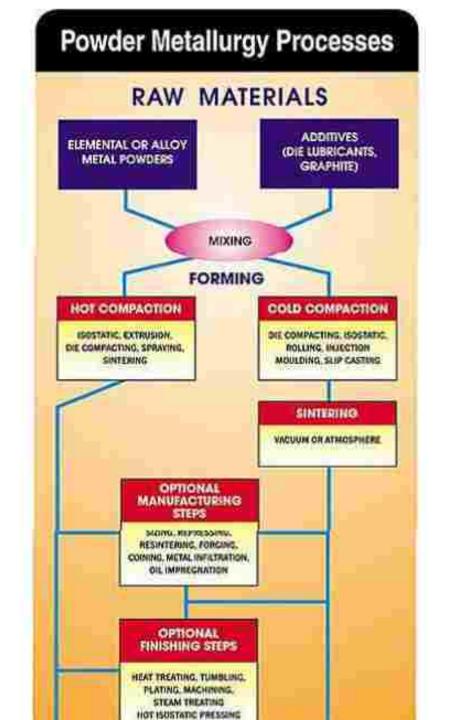












Production Sequence

- Procurement of raw materials needed for production of powdered metal part
- Mixing / Blending
- Compacting
- Sintering
- Post Treatments / Final Product
 - Post Treatments" include, but not limited to; coining, sizing, closed forging, copper infiltration, machining, heat treatment, induction hardening, stream treatment, tumbling (deburring), shot blasting, shot peening, oil impregnation, and plating.

P/M Sintering Process



P/M Material

Blending

Compacting

Sintering

Product